

## Evaluation of mechanical threshold for alloys with MatCalc (rel. 6.04.1004)

P. Warczok



# Mechanical threshold

- Yield stress at temperature 0 K
- Microstructure dependent

$$\sigma_0 = f \left( \rho, d_g, d_{sg}, c_i^m, r_{p_j}, N_{p_j}, \dots \right)$$

- Thermal & athermal contributions

$$\sigma_0 = \sigma_{ath} + \sigma_{th}$$

$\rho$  - Dislocation density [m<sup>-2</sup>]

$d_g$  - Grain diameter [m]

$d_{sg}$  - Subgrain diameter [m]

$c_i^m$  - Concentration of i-element in matrix

$r_{p_j}$  - Radius of precipitate j [m]

$N_{p_j}$  - Number density of precipitate j [m<sup>-3</sup>]

$\sigma_{ath}$  - Athermal contribution [Pa]

$\sigma_{th}$  - Thermal contribution [Pa]

# Model overview

- Contributions to mechanical threshold,  $\sigma_0$ 
  - Intrinsic strength,  $\sigma_i$
  - Work hardening,  $\sigma_{disl}$
  - Grain/subgrain boundary strengthening,  $\sigma_{gb}$ ,  $\sigma_{sgb}$
  - Solid solution strengthening,  $\sigma_{ss}$
  - Precipitation strengthening,  $\sigma_{prec}$

$$\sigma_0 = f(\sigma_i, \sigma_{disl}, \sigma_{gb}, \sigma_{sgb}, \sigma_{ss}, \sigma_{prec})$$

# Model overview

- Contributions to mechanical threshold,  $\sigma_0$ 
  - Intrinsic strength,  $\sigma_i$
  - Work hardening,  $\sigma_{disl}$
  - Grain/subgrain boundary strengthening,  $\sigma_{gb}$ ,  $\sigma_{sgb}$
  - Solid solution strengthening,  $\sigma_{ss}$
  - Precipitation strengthening,  $\sigma_{prec}$

$$\sigma_0 = f(\sigma_i, \sigma_{disl}, \sigma_{gb}, \sigma_{sgb}, \sigma_{ss}, \sigma_{prec})$$

# Intrinsic strength, $\sigma_i$

Precipitation domains ...

Precipitation domains ...

Ni\_matrix\*

General Mech. Props MS Evolution Trapping Special

General Solid Solution Segregation CC Diffusion Precipitation

Mechanical properties ...

Young's Modulus [Pa] (222750-83.6\*T\$C)\*1e6

Taylor factor (2.5-3.1) 3.06 Poisson's ratio 0.3

Speed of sound 5100.0

Matrix strength evaluation ...

Basic strength [Pa] 20.0e6

Hall-Petch coeff (gb/sgb) 0.74e6 / 0.0e6

Disl. strengt. coeff. (a1/a2) 0.5 / 0.0

Dynamic strength ...

delta\_F\_lt\_fact 1.0 delta\_F\_ht 0.0 coupl.-exp 3.0

eps\_dot\_fact 1.0 exp\_ht 1.0/3.0

Total strength coupling coefficients ...

Coeff. thermal + athermal (1.0) 1.0

Cancel OK

variables ...

variables	value
▲ kinetics: pd strength	
▲ TYSBS*	
TYSBSnickelmatrix	2.18e+07

category: kinetics: pd strength  
expression: TYSBS\$\*  
legal unit qualifiers: \*none\*  
-> basic yield strength of precipitation domain

# Model overview

- Contributions to yield strength,  $\sigma_{YS}$ 
  - Intrinsic strength,  $\sigma_i$
  - Work hardening,  $\sigma_{disl}$
  - Grain/subgrain boundary strengthening,  $\sigma_{gb}$ ,  $\sigma_{sgb}$
  - Solid solution strengthening,  $\sigma_{ss}$
  - Precipitation strengthening,  $\sigma_{prec}$

$$\sigma_0 = f(\sigma_i, \sigma_{disl}, \sigma_{gb}, \sigma_{sgb}, \sigma_{ss}, \sigma_{prec})$$

# Work hardening, $\sigma_{disl}$

- Taylor equation

$$\sigma_{disl} = \alpha M G b \sqrt{\rho}$$

$M$  - Taylor factor

$G$  - Shear modulus

$b$  - Burger's vector

$\rho$  - Dislocation density

$\alpha$  - Strengthening coefficient

# Work hardening, $\sigma_{disl}$

- Taylor equation

$$\sigma_{disl} = \alpha M G b \sqrt{\rho}$$

$M$  - Taylor factor

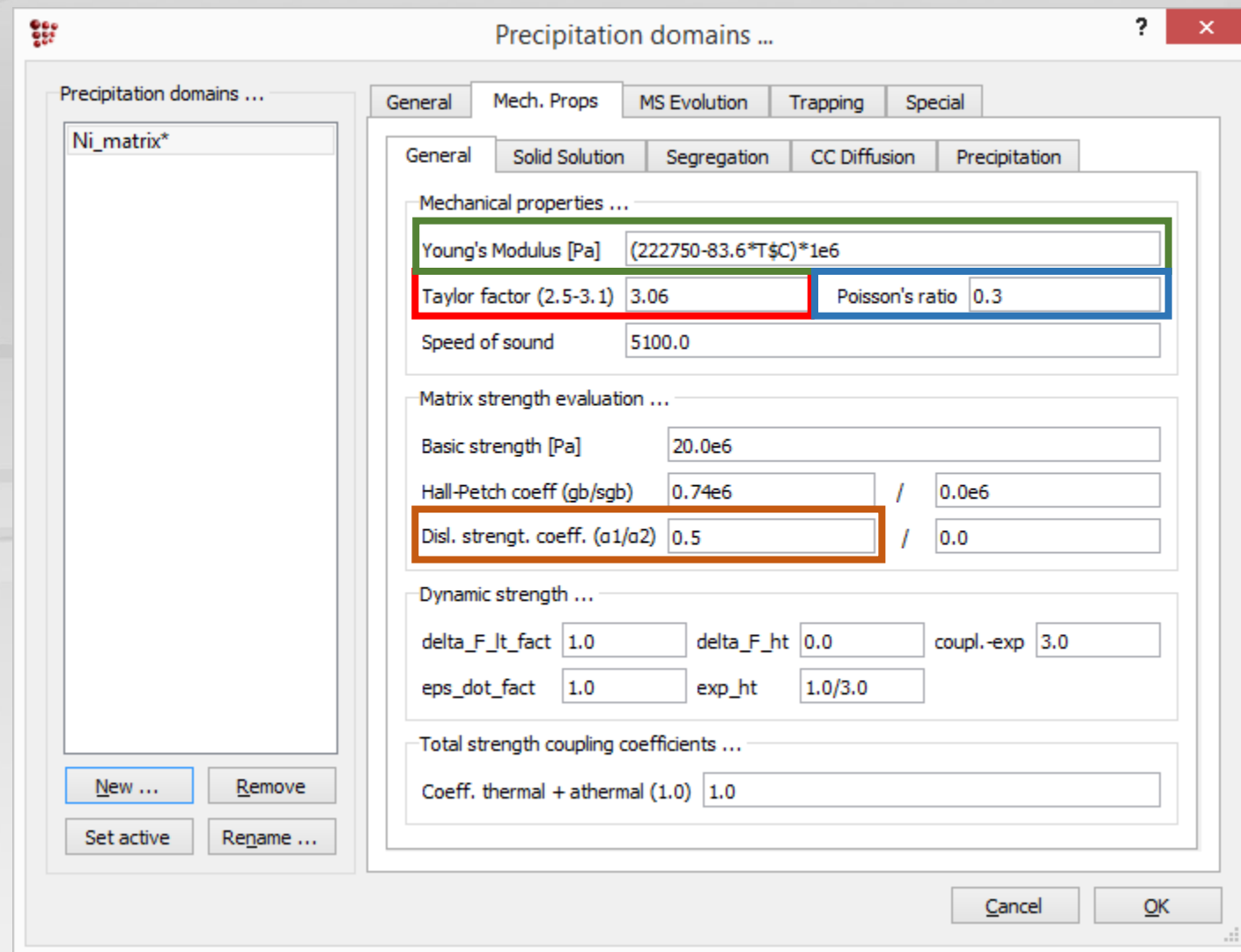
$G$  - Shear modulus

$b$  - Burger's vector

$\rho$  - Dislocation density

$\alpha$  - Strengthening coefficient

$$G = \frac{E}{2(1-\nu)}$$



# Work hardening, $\sigma_{disl}$

- Taylor equation

$$\sigma_{disl} = \alpha M G b \sqrt{\rho}$$

$M$  - Taylor factor

$G$  - Shear modulus

$b$  - Burger's vector

$\rho$  - Dislocation density

$\alpha$  - Strengthening coefficient

Precipitation domains ...

Precipitation domains ...

Ni\_matrix\*

General | Mech. Props | MS Evolution | Trapping | Special

Thermodynamic matrix phase ...  
FCC\_A1

Microstructure parameters ...

equilibrium dislocation density [m <sup>-2</sup> ]	1.0e11
initial grain diameter [m]	100.0e-6
elongation factor	1
initial subgrain diameter [m]	100.0e-6
elongation factor	1

Burger's vector

automatic      manual value [m] 2.5e-10

Dislocation density is evaluated from the relevant MS Evolution model. If the model is inactive the initial value as depicted here is taken

Cancel OK

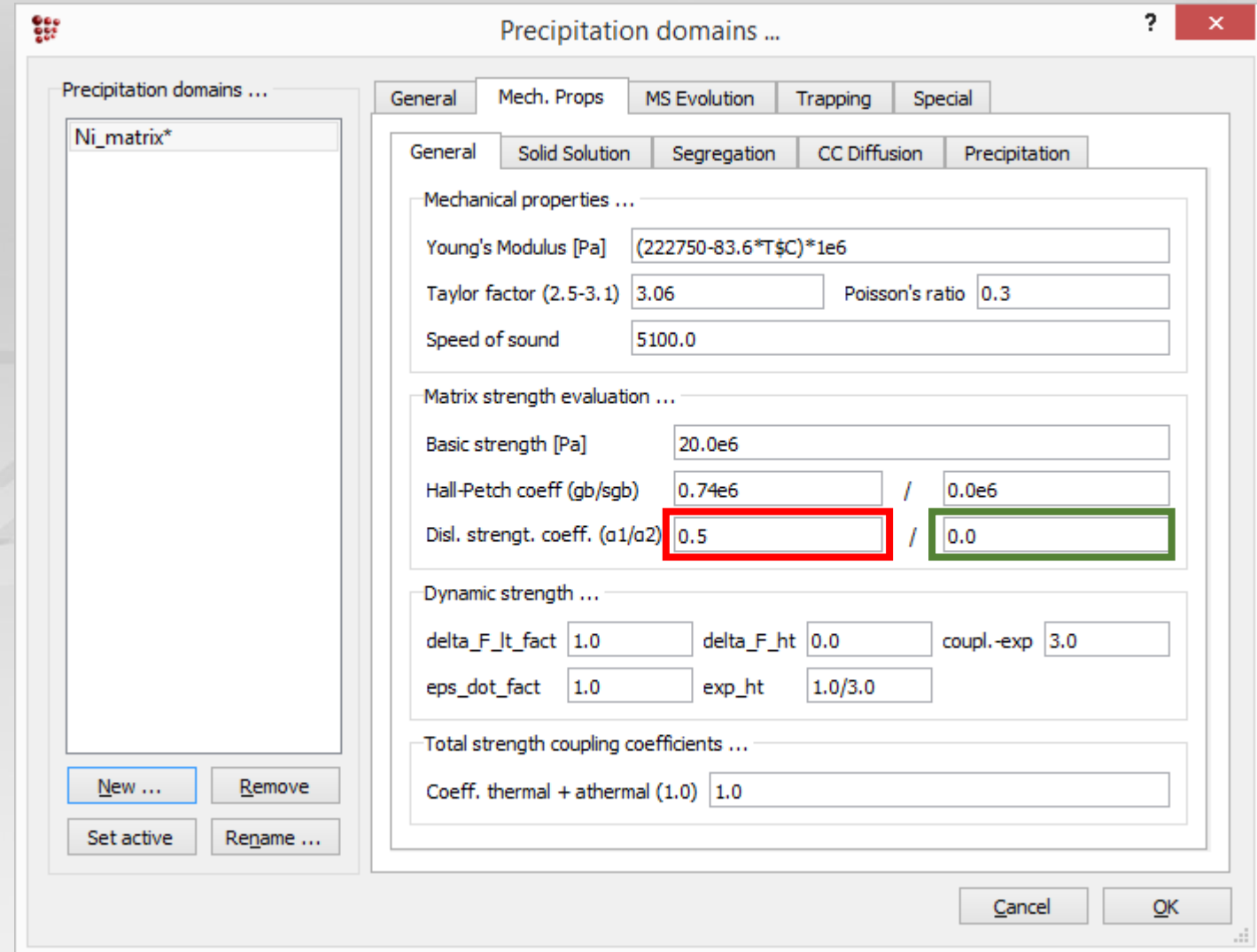
# Work hardening, $\sigma_{disl}$

- Taylor equation
  - Two parameter model

$$\sigma_{disl} = \alpha_1 M G b \sqrt{\rho_1} + \alpha_2 M G b \sqrt{\rho_2}$$

$\rho_1$  - Internal dislocation density

$\rho_2$  - Wall dislocation density



# Work hardening, $\sigma_{disl}$

- Taylor equation
  - Two parameter model

$$\sigma_{disl} = \alpha_1 M G b \sqrt{\rho_1} + \alpha_2 M G b \sqrt{\rho_2}$$

$\rho_1$  - Internal dislocation density

$\rho_2$  - Wall dislocation density

variables	value
▾ kinetics: pd strength	
▾ TDSS*	
TDS\$nickelmatrix	7.1404e+06

category: kinetics: pd strength  
 expression: TDS\$nickelmatrix  
 legal unit qualifiers: \*none\*  
 -> dislocation yield strength contribution in precipitation domain

# Model overview

- Contributions to yield strength,  $\sigma_{YS}$ 
  - Intrinsic strength,  $\sigma_i$
  - Work hardening,  $\sigma_{disl}$
  - Grain/subgrain boundary strengthening,  $\sigma_{gb}$ ,  $\sigma_{sgb}$
  - Solid solution strengthening,  $\sigma_{ss}$
  - Precipitation strengthening,  $\sigma_{prec}$

$$\sigma_0 = f(\sigma_i, \sigma_{disl}, \sigma_{gb}, \sigma_{sgb}, \sigma_{ss}, \sigma_{prec})$$

# Grain/subgrain boundary strengthening, $\sigma_{gb}$ , $\sigma_{sgb}$

- Hall-Petch equation

$$\sigma_{gb} = \frac{k_{gb}}{\sqrt{D}} \quad \sigma_{sgb} = \frac{k_{sgb}}{\sqrt{\delta}}$$

$D$  - Grain diameter

$\delta$  - Subgrain diameter

$k_n$  - Constant

# Grain/subgrain boundary strengthening, $\sigma_{gb}$ , $\sigma_{sgb}$

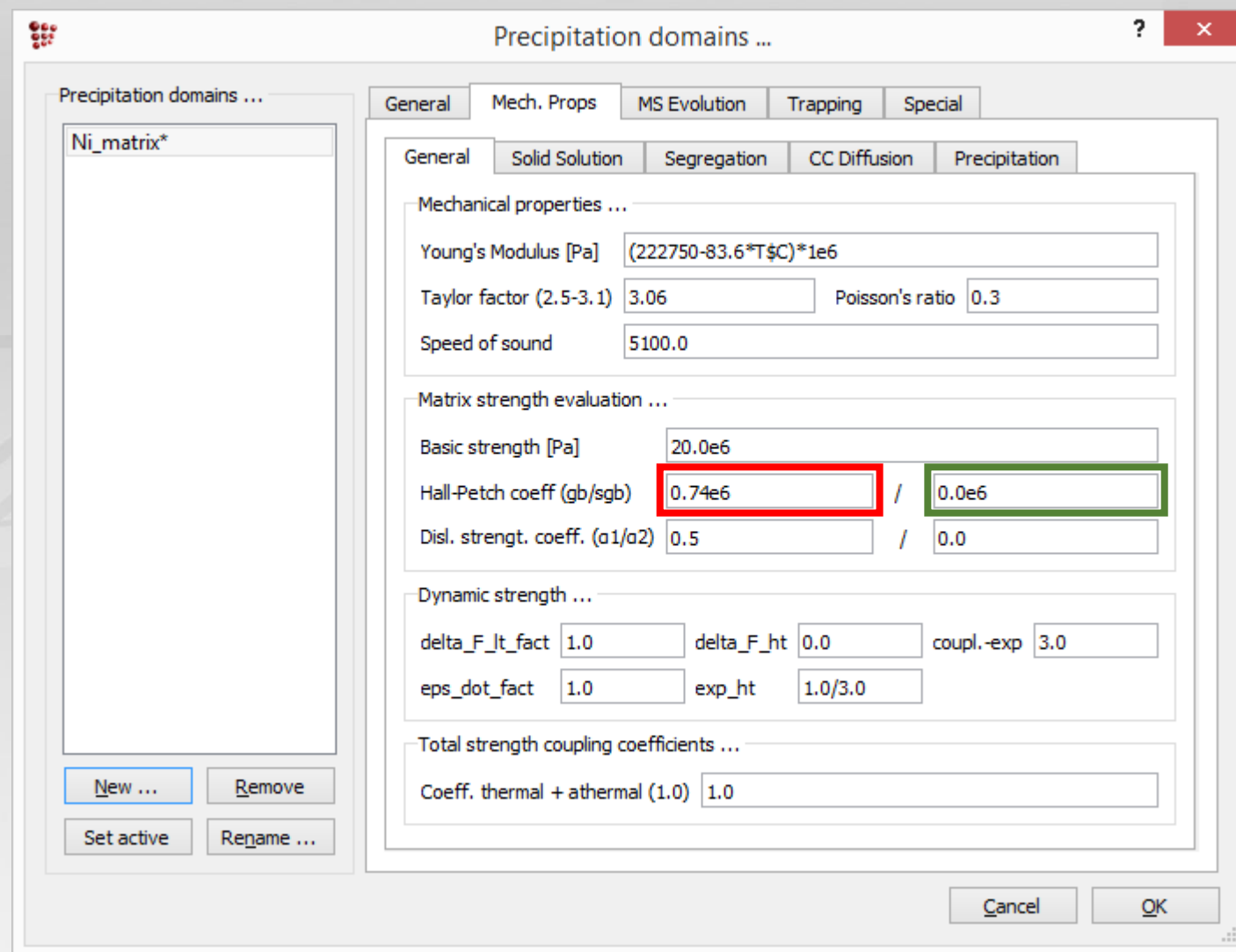
- Hall-Petch equation

$$\sigma_{gb} = \frac{k_{gb}}{\sqrt{D}} \quad \sigma_{sgb} = \frac{k_{sgb}}{\sqrt{\delta}}$$

$D$  - Grain diameter

$\delta$  - Subgrain diameter

$k_n$  - Constant



# Grain/subgrain boundary strengthening, $\sigma_{gb}$ , $\sigma_{sgb}$

- Hall-Petch equation

$$\sigma_{gb} = \frac{k_{gb}}{\sqrt{D}} \quad \sigma_{sgb} = \frac{k_{sgb}}{\sqrt{\delta}}$$

$D$  - Grain diameter

$\delta$  - Subgrain diameter

$k_n$  - Constant

Precipitation domains ...

General | Mech. Props | MS Evolution | Trapping | Special

Thermodynamic matrix phase ...  
FCC\_A1

Microstructure parameters ...

equilibrium dislocation density [m-2] 1.0e11

initial grain diameter [m] 100.0e-6 elongation factor 1

initial subgrain diameter [m] 100.0e-6 elongation factor 1

Burger's vector

automatic manual value [m] 2.5e-10

Diameters are evaluated from the relevant MS Evolution models. If models are inactive the initial values as depicted here are taken

Cancel OK

# Grain/subgrain boundary strengthening, $\sigma_{gb}$ , $\sigma_{sgb}$

- Hall-Petch equation

$$\sigma_{gb} = \frac{k_{gb}}{\sqrt{D}} \quad \sigma_{sgb} = \frac{k_{sgb}}{\sqrt{\delta}}$$

$D$  - Grain diameter

$\delta$  - Subgrain diameter

$k_n$  - Constant

variables	value
kinetics: pd strength	
kinetics: pd strength	
TGSS*	
TGSSnickelmatrix	3.57771e+07

category: kinetics: pd strength  
expression: TGS\$\*  
legal unit qualifiers: \*none\*  
-> fine grain yield strength contribution in precipitation domain

variables	value
kinetics: pd strength	
kinetics: pd strength	
TSGS*	
TSGSnickelmatrix	0

category: kinetics: pd strength  
expression: TSGS\$nickelmatrix  
legal unit qualifiers: \*none\*  
-> subgrain yield strength contribution in precipitation domain

# Model overview

- Contributions to yield strength,  $\sigma_{YS}$ 
  - Intrinsic strength,  $\sigma_i$
  - Work hardening,  $\sigma_{disl}$
  - Grain/subgrain boundary strengthening,  $\sigma_{gb}$ ,  $\sigma_{sgb}$
  - Solid solution strengthening,  $\sigma_{ss}$
  - Precipitation strengthening,  $\sigma_{prec}$

$$\sigma_0 = f(\sigma_i, \sigma_{disl}, \sigma_{gb}, \sigma_{sgb}, \sigma_{ss}, \sigma_{prec})$$

# Solid solution strengthening, $\sigma_{SS}$

$$\sigma_{SS} = \left[ \left( \sum_i \left( k_i c_i^{n_i} \right)^{m_{sub}} \right)_{sub} + \left( \sum_i \left( k_i c_i^{n_i} \right)^{m_{int}} \right)_{int} \right]^{\frac{1}{m_{tot}}}$$

$k_i$  - Coefficient for element i

$c_i$  - Element i content in the prec. Domain  
(mole fraction)

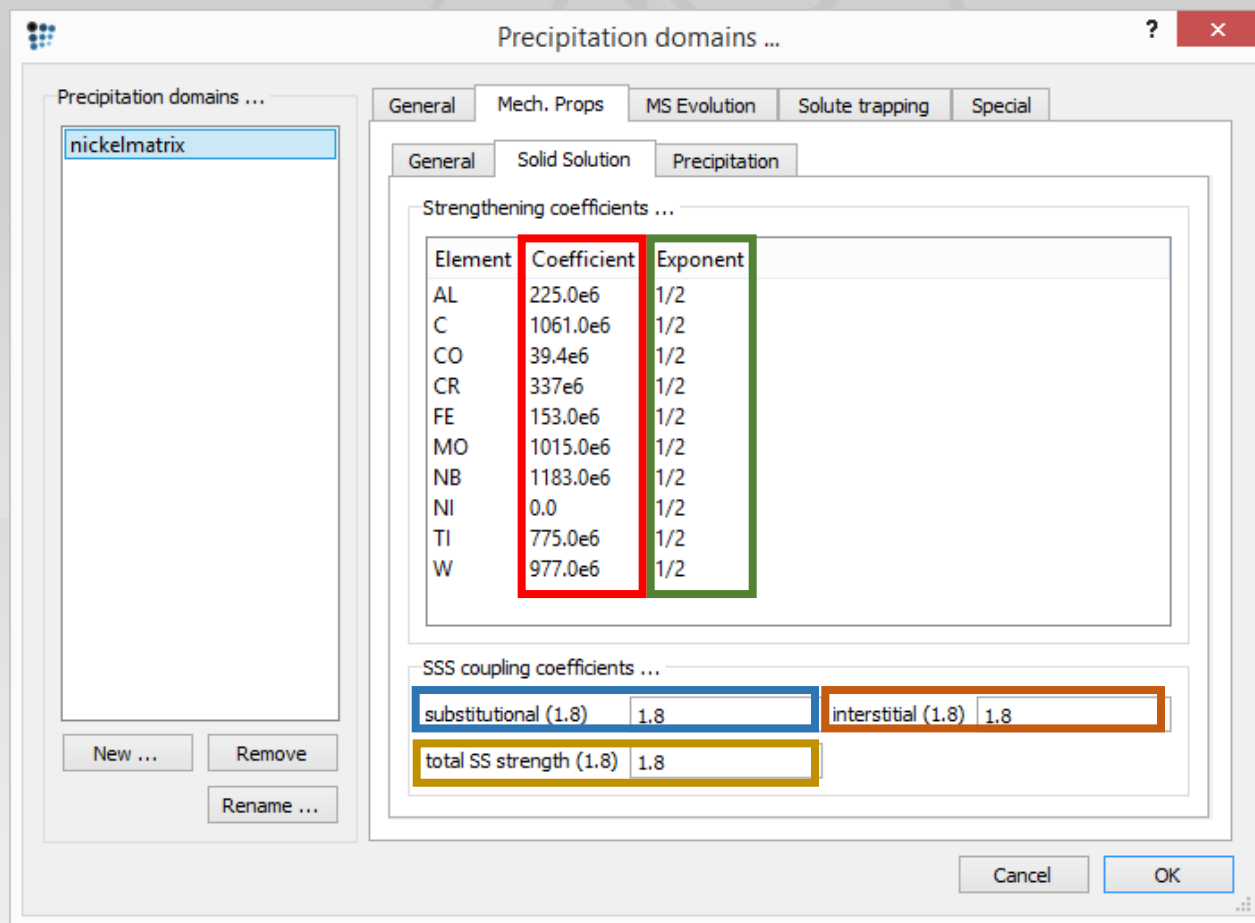
$n_i$  - Exponent for element i

$m_{sub}$  - Exponent for substitutional elements

$m_{int}$  - Exponent for interstitial elements

$m_{tot}$  - Global exponent

# Solid solution strengthening, $\sigma_{SS}$



$$\sigma_{SS} = \left[ \left( \sum_i (k_i c_i^{n_i})^{m_{sub}} \right)_{sub} + \left( \sum_i (k_i c_i^{n_i})^{m_{int}} \right)_{int} \right]^{\frac{1}{m_{tot}}}$$

$k_i$  - Coefficient for element  $i$

$c_i$  - Element  $i$  content in the prec. domain

$n_i$  - Exponent for element  $i$

$m_{sub}$  - Exponent for substitutional elements

$m_{int}$  - Exponent for interstitial elements

$m_{tot}$  - Global exponent

# Solid solution strengthening, $\sigma_{SS}$

$$\sigma_{SS} = \left[ \left( \sum_i (k_i c_i^{n_i})^{m_{sub}} \right)_{sub}^{\frac{m_{tot}}{m_{sub}}} + \left( \sum_i (k_i c_i^n)^{m_{int}} \right)_{int}^{\frac{m_{tot}}{m_{int}}} \right]^{\frac{1}{m_{tot}}}$$

variables	value
▲ kinetics: pd strength	
▲ TSSSS*	
TSSSSnickelmatrix	2.82649e+08

category: kinetics: pd strength  
 expression: TSSSS\$\*  
 legal unit qualifiers: \*none\*  
 -> solid solution yield strength contribution in precipitation domain

# Solid solution strengthening, $\sigma_{SS}$

- Solid solution strengthening,  $\sigma_{SS}$

$$\sigma_{SS} = \left[ \left( \sum_i \left( k_i c_i^{n_i} \right)^{m_{sub}} \right)^{\frac{m_{tot}}{m_{sub}}} + \left( \sum_i \left( k_i c_i^{n_i} \right)^{m_{int}} \right)^{\frac{m_{tot}}{m_{int}}} \right]^{\frac{1}{m_{tot}}}$$

variables	value
kinetics: pd strength	
TSSS_EL\$*\$*	
TSSS_EL\$nickelmatrix\$*	
TSSS_EL\$nickelmatrix\$VA	0
TSSS_EL\$nickelmatrix\$AL	2.78874e+07
TSSS_EL\$nickelmatrix\$C	4.56667e+07
TSSS_EL\$nickelmatrix\$CO	1.27085e+07
TSSS_EL\$nickelmatrix\$CR	1.70234e+08
TSSS_EL\$nickelmatrix\$FE	5.49895e+07
TSSS_EL\$nickelmatrix\$MO	1.47716e+08
TSSS_EL\$nickelmatrix\$NB	8.44694e+07
TSSS_EL\$nickelmatrix\$NI	0
TSSS_EL\$nickelmatrix\$TI	1.95807e+07
TSSS_EL\$nickelmatrix\$W	5.71824e+07

category: kinetics: pd strength  
 expression: TSSS\_EL\$\*\$W  
 legal unit qualifiers: \*none\*  
 -> solid solution yield strength contribution of element in precipitation domain

# Model overview

- Contributions to yield strength,  $\sigma_{YS}$ 
  - Intrinsic strength,  $\sigma_i$
  - Work hardening,  $\sigma_{disl}$
  - Grain/subgrain boundary strengthening,  $\sigma_{gb}$ ,  $\sigma_{sgb}$
  - Solid solution strengthening,  $\sigma_{ss}$
  - Precipitation strengthening,  $\sigma_{prec}$

$$\sigma_0 = f(\sigma_i, \sigma_{disl}, \sigma_{gb}, \sigma_{sgb}, \sigma_{ss}, \sigma_{prec})$$

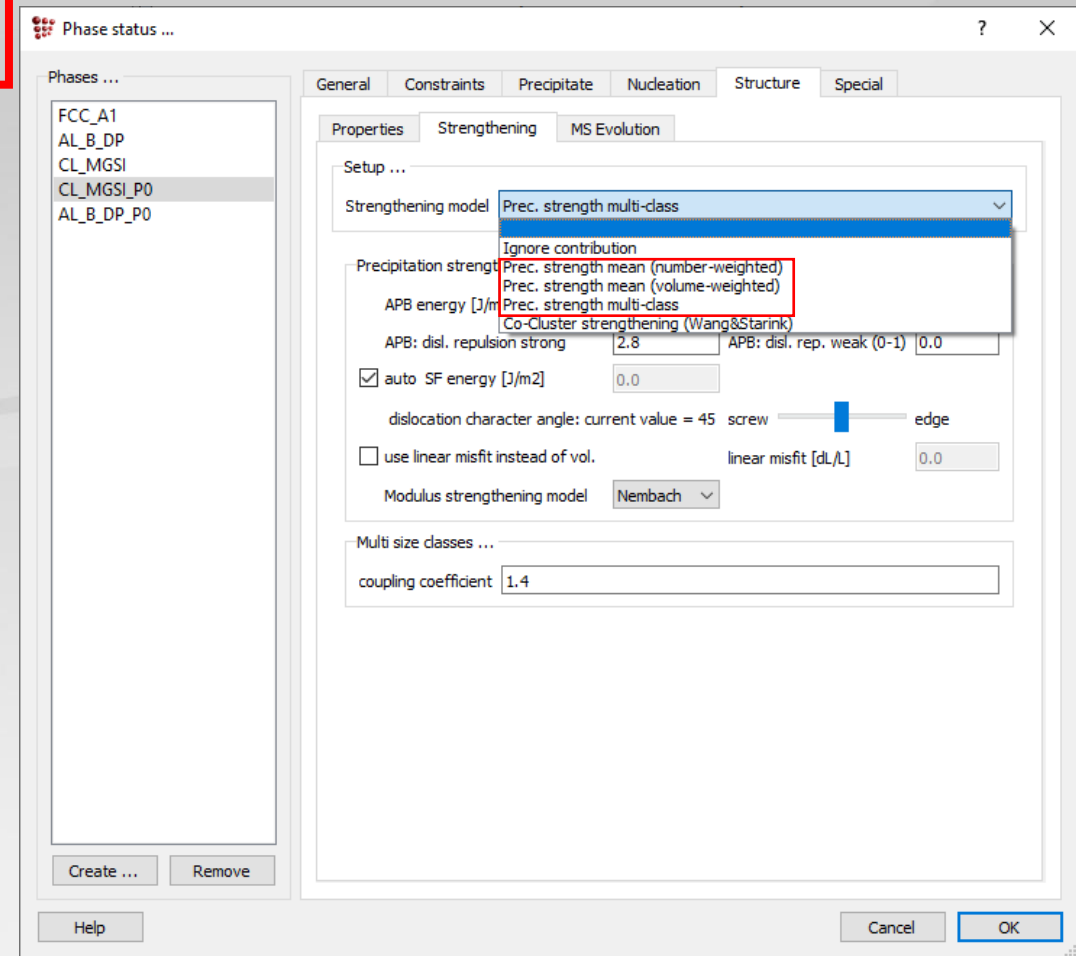
# Precipitation strengthening, $\sigma_{prec}$

- 2 alternative models available
  - Size distribution dependent strengthening
  - Co-cluster strengthening
- $\tau_{prec} \rightarrow \sigma_{prec}$

$\tau_{prec}$  - Critical shear stress for a dislocation to cut/by-pass a particle

# Precipitation strengthening, $\sigma_{prec}$

- 2 alternative models available
- Size distribution dependent strengthening
- Co-cluster strengthening



# Size distribution dependent strengthening

- Precipitate size dependence
- Some general parameters/settings
- 2 scenarios for dislocation behavior:
  - Non-shearable particles (Orowan mechanism) → bypassing precipitate
  - Shearable particles (weak or strong) → cutting precipitate
- Critical stresses for both scenarios are evaluated

# Size distribution dependent strengthening

- Precipitate size dependence
- Some general parameters/settings
- 2 scenarios for dislocation behavior:
  - Non-shearable particles (Orowan mechanism) → bypassing precipitate
  - Shearable particles (weak or strong) → cutting precipitate
- Critical stresses for both scenarios are evaluated

# Size distribution dependent strengthening

- Precipitate size dependence
  - Contributions dependent on precipitate size
  - Various choices for precipitate size parameter selection possible
    - Number weighted mean radius
    - Volume weighted mean radius
    - Size class radius

# Size distribution dependent strengthening

- Various choices possible
  - Number-weighted mean radius,  $r_{m,n}$

$$r_{m,n} = \frac{\sum_i N_i r_i}{\sum_i N_i}$$

- Volume-weighted mean radius,  $r_{m,v}$

$$r_{m,v} = \frac{\sum_i N_i r_i^4}{\sum_i N_i r_i^3}$$

- Size class radius,  $r_i$

Precipitate size distribution

Size class index $i$	Size class radius $r_i$	Size class number density $N_i$
0	$r_0$	$N_0$
1	$r_1$	$N_1$
2	$r_2$	$N_2$
3	$r_3$	$N_3$
...	...	...

# Size distribution dependent strengthening

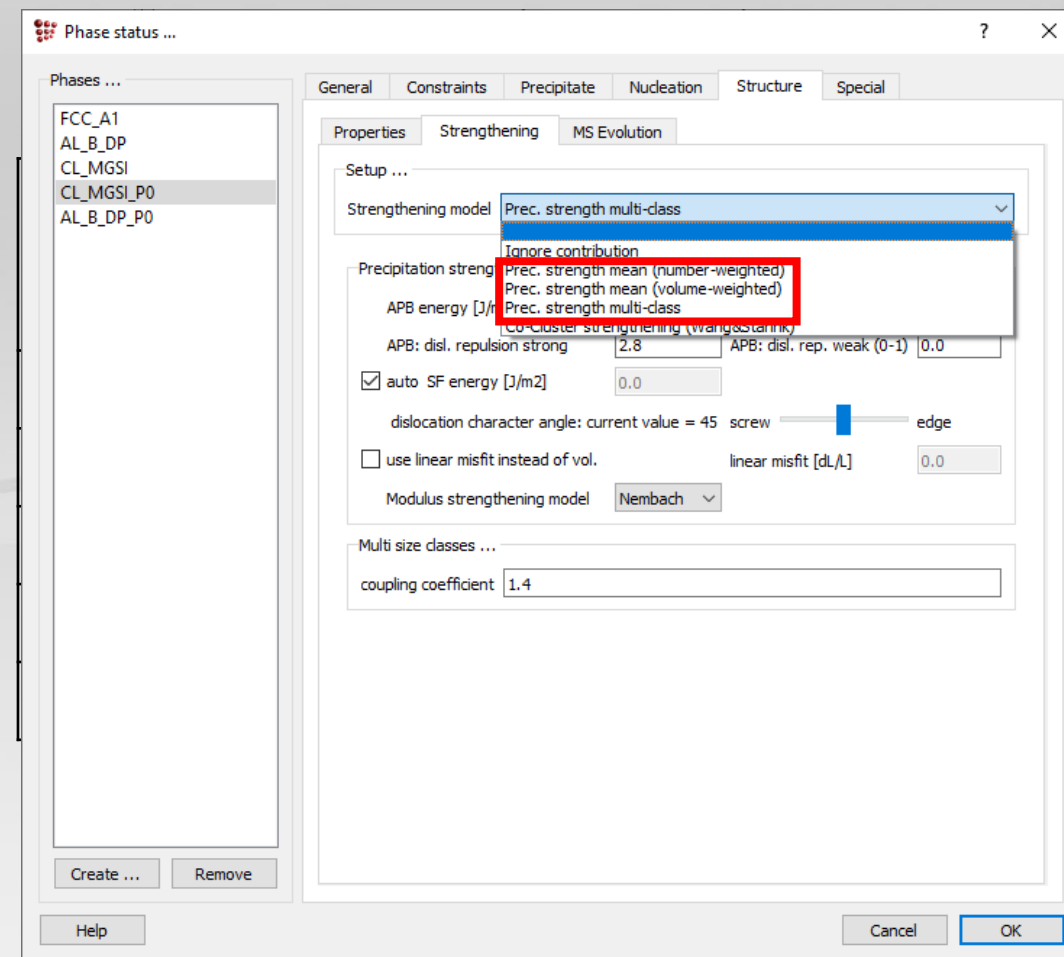
- Various choices possible
  - Number-weighted mean radius,  $r_{m,n}$

$$r_{m,n} = \frac{\sum_i N_i r_i}{\sum_i N_i}$$

- Volume-weighted mean radius,  $r_{m,v}$

$$r_{m,v} = \frac{\sum_i N_i r_i^4}{\sum_i N_i r_i^3}$$

- Size class radius,  $r_i$  (multi-class model)



# Size distribution dependent strengthening

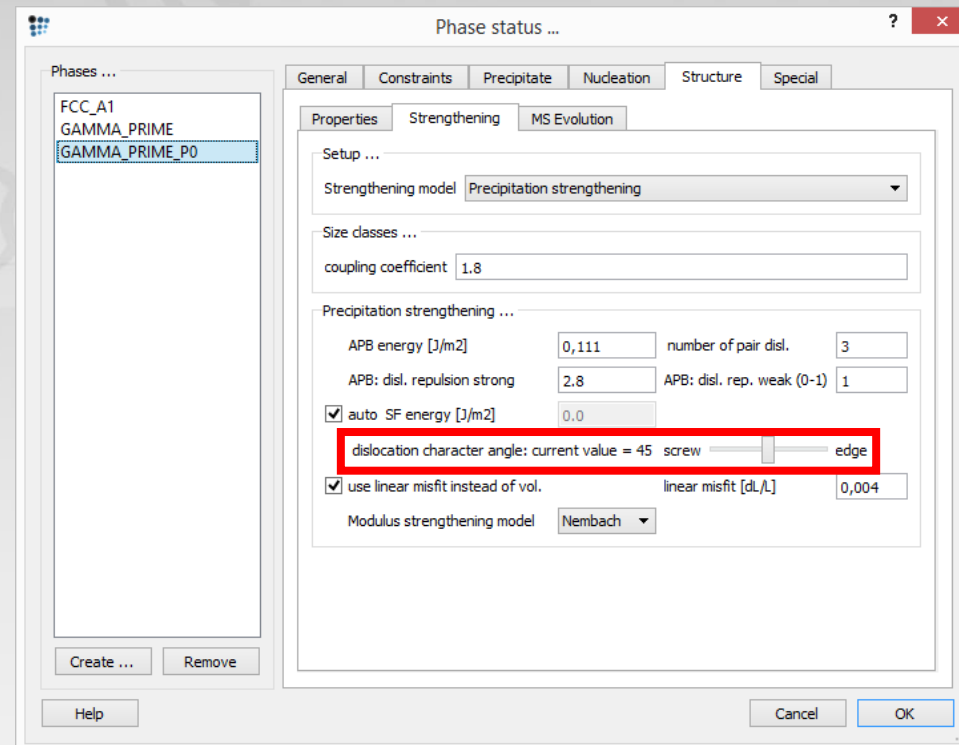
- Precipitate size dependence
- Some general parameters/settings
- 2 scenarios for dislocation behavior:
  - Non-shearable particles (Orowan mechanism) → bypassing precipitate
  - Shearable particles (weak or strong) → cutting precipitate
- Critical stresses for both scenarios are evaluated

# Size distribution dependent strengthening

- Some general parameters/settings
  - Angle between dislocation line and Burger's vector,  $\theta$   
(edge/screw ratio;  $\theta = 0$  for pure screw;  $\theta = \pi/2$  for pure edge)
  - Equivalent radius,  $r_{eq}$  (describes precipitate-dislocation interference area)
  - Mean distance between the precipitate surfaces,  $L_S$

# Size distribution dependent strengthening

- Some general parameters/settings
  - Angle between dislocation line and Burger's vector  $\theta$   
(edge/screw ratio;  $\theta = 0$  for pure screw;  $\theta = \pi/2$  for pure edge)



# Size distribution dependent strengthening

- Some general parameters/settings
  - Equivalent radius,  $r_{eq}$  (describes precipitate-dislocation interference area)

$$r_{eq} = \frac{\pi}{4} r_m$$

$r_m$  - Precipitate mean radius

# Size distribution dependent strengthening

- Some general parameters/settings
  - Mean distance between the precipitate surfaces,  $L_S$

$$L_S = \sqrt{\frac{\ln 3}{2\pi \sum_{class} N_{V,class} r_{m,class}^2} + 4r_{ss}^2} - 2r_{ss}$$

$$r_{ss} = \sqrt{\frac{2 \sum_{class} N_{V,class} r_{m,class}^2}{3 \sum_{class} N_{V,class} r_{m,class}}}$$

$N_{V,class}$  - Precipitate number density within the class

$r_{m,class}$  - Precipitate mean radius within the class

variables	value
▾ kinetics: precipitates	
▾ L_MEAN_2DS*	
L_MEAN_2DSGAMMA_PRIME_P0	2.02552e-08

category: kinetics: precipitates  
 expression: L\_MEAN\_2DSGAMMA\_PRIME\_P0  
 legal unit qualifiers: \*none\*  
 -> mean distance between randomly distributed precipitates on a single plane (2-dimensional)

# Size distribution dependent strengthening

- Precipitate size dependence
- Some general parameters/settings
- 2 scenarios for dislocation behavior:
  - Non-shearable particles (Orowan mechanism) → bypassing precipitate
  - Shearable particles (weak or strong) → cutting precipitate
- Critical stresses for both scenarios are evaluated

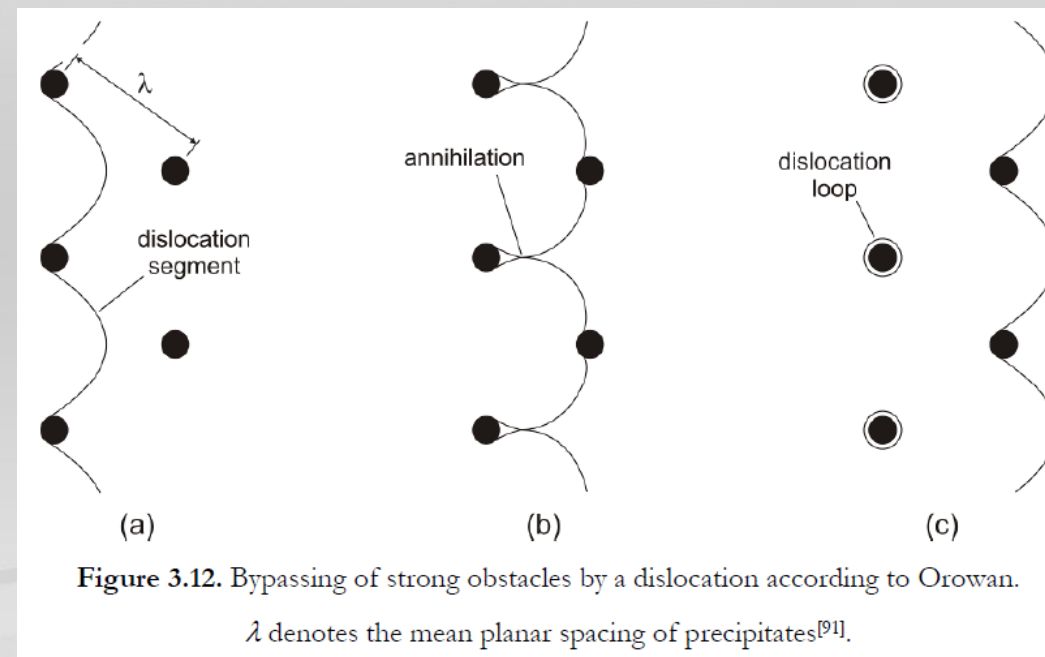
# Size distribution dependent strengthening

- Precipitate size dependence
- Some general parameters/settings
- 2 scenarios for dislocation behavior:
  - Non-shearable particles (Orowan mechanism) → bypassing precipitate
  - Shearable particles (weak or strong) → cutting precipitate
- Critical stresses for both scenarios are evaluated

# Non-shearable particles

$$\tau_{nsh} = \frac{JGb}{2\pi L_S} \ln \left( \frac{\pi r_{eq}}{2 r_i} f(\theta, h) \right)$$

$$J = \frac{1 - \nu \left[ \cos^2 \left( \frac{\pi}{2} - \theta \right) \right]}{1 - \nu}$$



$$f(\theta, h) = \frac{h^{2/3}}{3} \left[ \left( \sqrt{\frac{3}{2+h^2}} + \sqrt{\frac{3}{h^2} + \frac{3}{2+h^2}} \right) \sin^2 \theta + \left( \sqrt{\frac{1}{h^2}} + \sqrt{\frac{9}{2+h^2} + \frac{1}{h^2}} \right) \cos^2 \theta \right]$$

# Non-shearable particles

$$\tau_{nsh} = \frac{JGb}{2\pi L_S} \ln \left( \frac{\pi r_{eq}}{2 r_i} f(\theta, h) \right)$$

$$J = \frac{1 - \nu \left[ \cos^2 \left( \frac{\pi}{2} - \theta \right) \right]}{1 - \nu}$$

$\tau_{nsh}$  - Critical stress for a dislocation to by-pass the precipitate

$G$  - Shear modulus

$b$  - Burgers vector

$\nu$  - Poisson's ratio

$r_{eq}$  - Equivalent radius

$\theta$  -  $\angle(b; \text{dislocation line})$

$r_i$  - dislocation core radius

$h$  - Shape factor

$$f(\theta, h) = \frac{h^{2/3}}{3} \left[ \left( \sqrt{\frac{3}{2+h^2}} + \sqrt{\frac{3}{h^2} + \frac{3}{2+h^2}} \right) \sin^2 \theta + \left( \sqrt{\frac{1}{h^2}} + \sqrt{\frac{9}{2+h^2} + \frac{1}{h^2}} \right) \cos^2 \theta \right]$$

# Non-shearable particles

$\tau_{nsh}$  - Critical stress for a dislocation to by-pass the precipitate

$G$  - Shear modulus

$b$  - Burgers vector

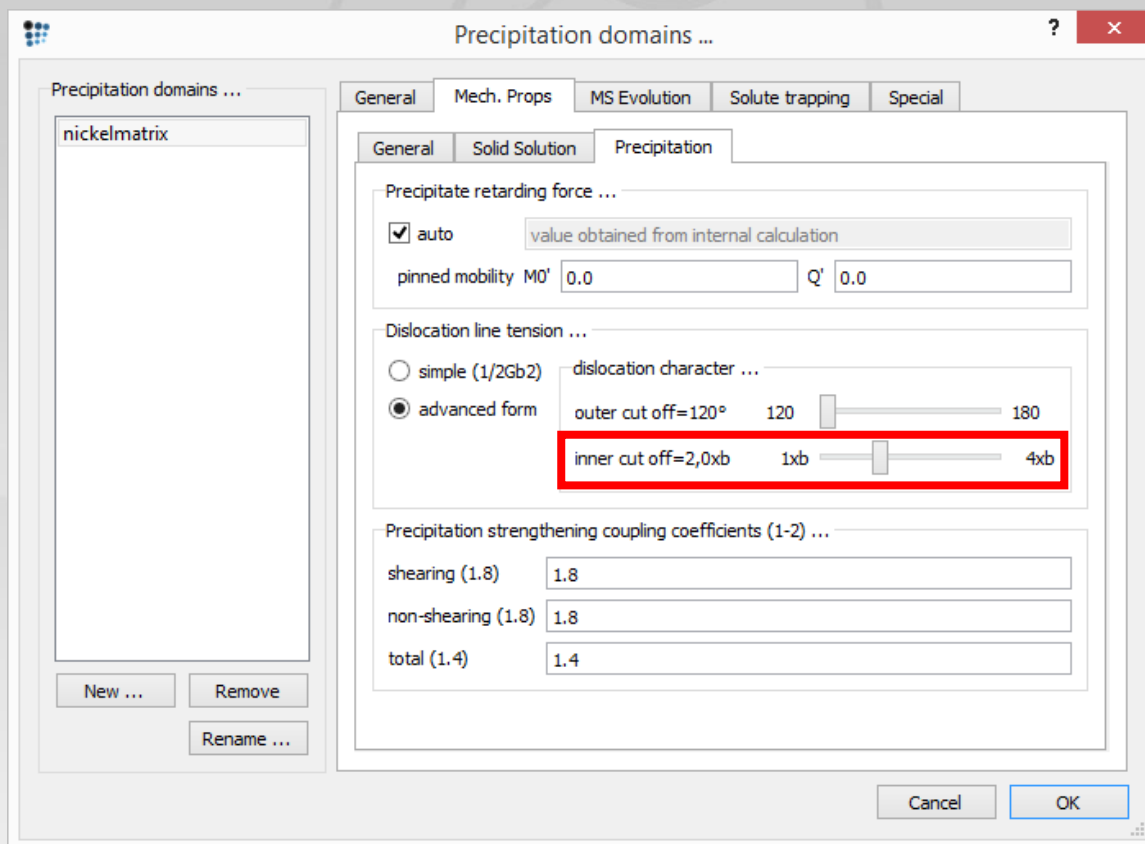
$\nu$  - Poisson's ratio

$r_{eq}$  - Equivalent radius

$\theta$  -  $\angle(b; \text{dislocation line})$

$r_i$  - dislocation core radius

$h$  - Shape factor



$$\left( \frac{1}{2} \right) \sin^2 \theta + \left( \sqrt{\frac{1}{h^2}} + \sqrt{\frac{9}{2 + h^2} + \frac{1}{h^2}} \right) \cos^2 \theta$$

# Non-shearable particles

$\tau_{nsh}$  - Critical stress for a dislocation to by-pass the precipitate

$G$  - Shear modulus

$b$  - Burgers vector

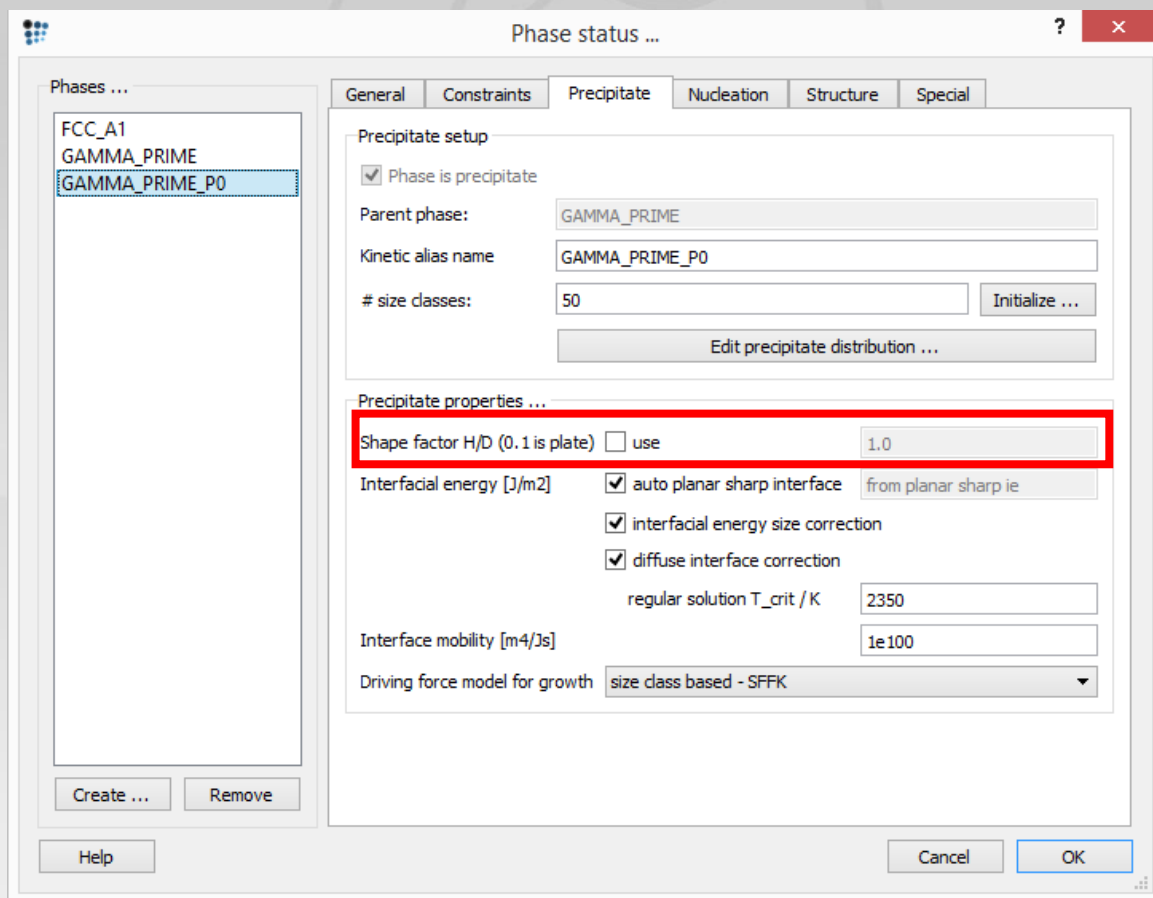
$\nu$  - Poisson's ratio

$r_{eq}$  - Equivalent radius

$\theta$  -  $\angle(b; \text{dislocation line})$

$r_i$  - dislocation core radius

$h$  - Shape factor



$$\left( \frac{1}{2} \right) \sin^2 \theta + \left( \sqrt{\frac{1}{h^2}} + \sqrt{\frac{9}{2 + h^2} + \frac{1}{h^2}} \right) \cos^2 \theta$$

# Non-shearable particles

$$\tau_{nsh} = \frac{JGb}{2\pi L_S} \ln \left( \frac{\pi r_{eq}}{2 r_i} f(\theta, h) \right)$$

$$J = \frac{1 - \nu \left[ \cos^2 \left( \frac{\pi}{2} - \theta \right) \right]}{1 - \nu}$$

$$f(\theta, h) = \frac{h^{2/3}}{3} \left[ \left( \sqrt{\frac{3}{2+h^2}} + \sqrt{\frac{3}{h^2} + \frac{3}{2+h^2}} \right) \sin^2 \theta + \left( \sqrt{\frac{1}{h^2}} + \sqrt{\frac{9}{2+h^2} + \frac{1}{h^2}} \right) \cos^2 \theta \right]$$

variables	value
kinetics: prec. strength	
TAO_OROWANS*	
TAO_OROWANS\$GAMMA_PRIME_P0	7.74823e+08

category: kinetics: prec. strength  
 expression: TAO\_OROWANS\$GAMMA\_PRIME\_P0  
 legal unit qualifiers: \*none\*  
 -> Ashby-Orowan shear stress for impenetrable precipitates of individual phase

# Size distribution dependent strengthening

- Precipitate size dependence
- Some general parameters/settings
- 2 scenarios for dislocation behavior:
  - Non-shearable particles (Orowan mechanism) → bypassing precipitate
  - Shearable particles (weak or strong) → cutting precipitate
- Critical stresses for both scenarios are evaluated

# Shearable particles

- Models for “weak” and “strong” particles
- Various effects (contributions) considered
  - Coherency effect
  - Modulus effect
  - Anti-phase boundary effect
  - Stacking fault effect
  - Interfacial effect

# Shearable particles

- Models for “weak” and “strong” particles
- Various effects (contributions) considered
  - Coherency effect
  - Modulus effect
  - Anti-phase boundary effect
  - Stacking fault effect
  - Interfacial effect

# „Weak“ vs. „strong“ particles

- Criterion: Dislocation bending angle  $\psi$  threshold
  - Strong resistance of particles  $\rightarrow$  High curvature of dislocation line  $\rightarrow$  small  $\psi$

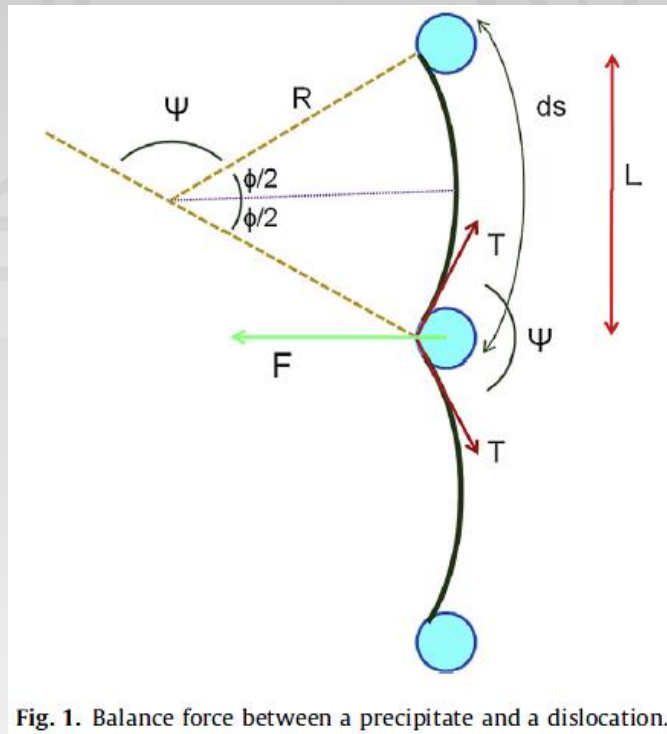


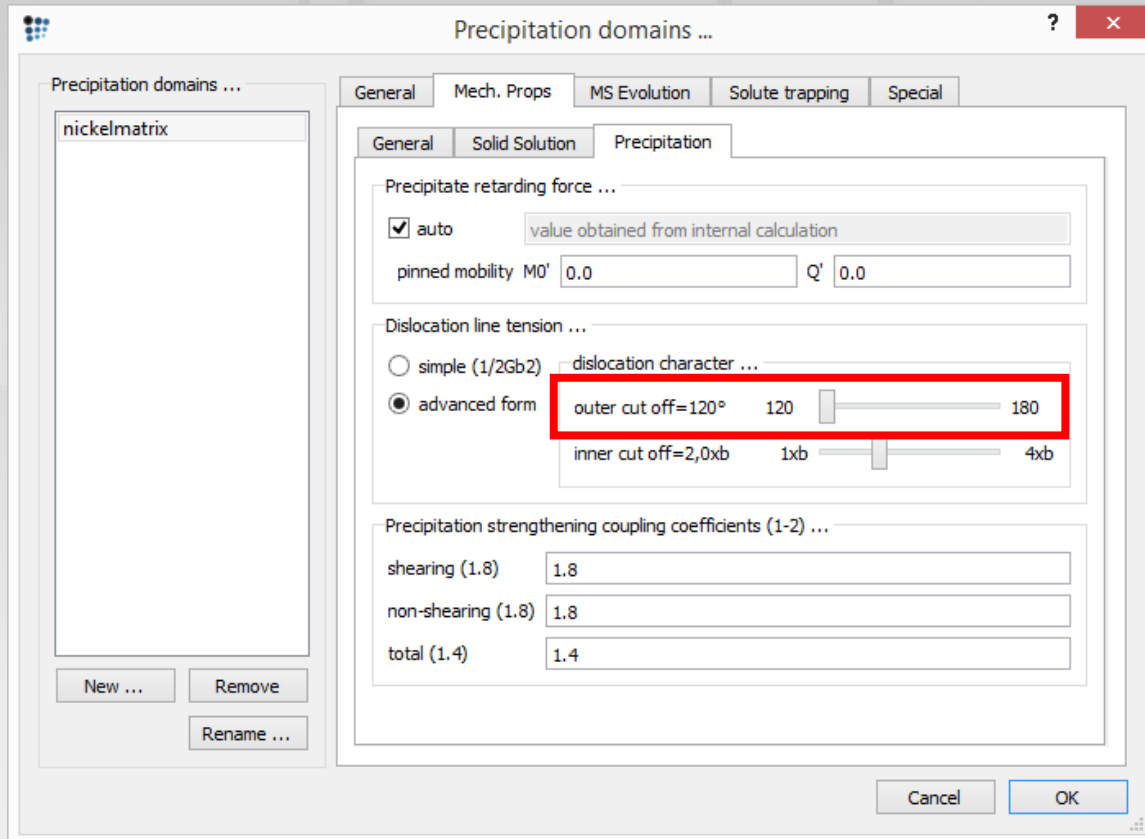
Fig. 1. Balance force between a precipitate and a dislocation.

$0^\circ - \psi \rightarrow$  “strong” particles

$\psi - 180^\circ \rightarrow$  “weak” particles

# „Weak“ vs. „strong“ particles

- Criterion: **Dislocation bending angle  $\psi$  threshold**
- Strong resistance of particles  $\rightarrow$  High curvature of dislocation line  $\rightarrow$  small  $\psi$

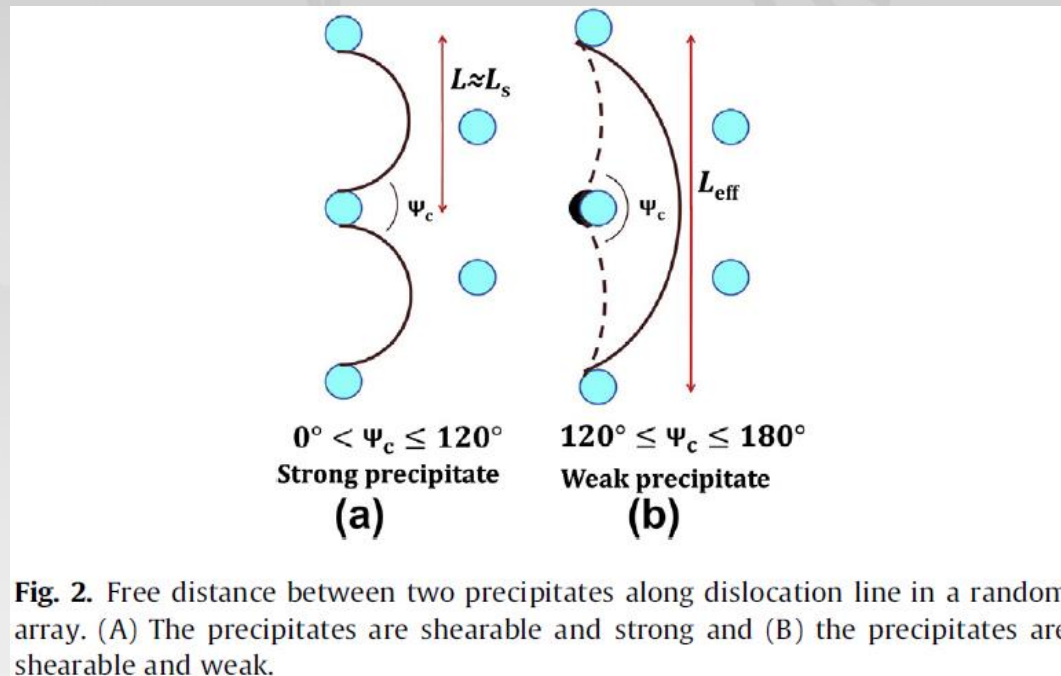


$0^\circ - \psi \rightarrow$  “strong” particles

$\psi - 180^\circ \rightarrow$  “weak” particles

# „Weak“ vs. „strong“ particles

- Criterion: Dislocation bending angle  $\psi$  threshold
  - Strong resistance of particles  $\rightarrow$  High curvature of dislocation line  $\rightarrow$  small  $\psi$



**Fig. 2.** Free distance between two precipitates along dislocation line in a random array. (A) The precipitates are shearable and strong and (B) the precipitates are shearable and weak.

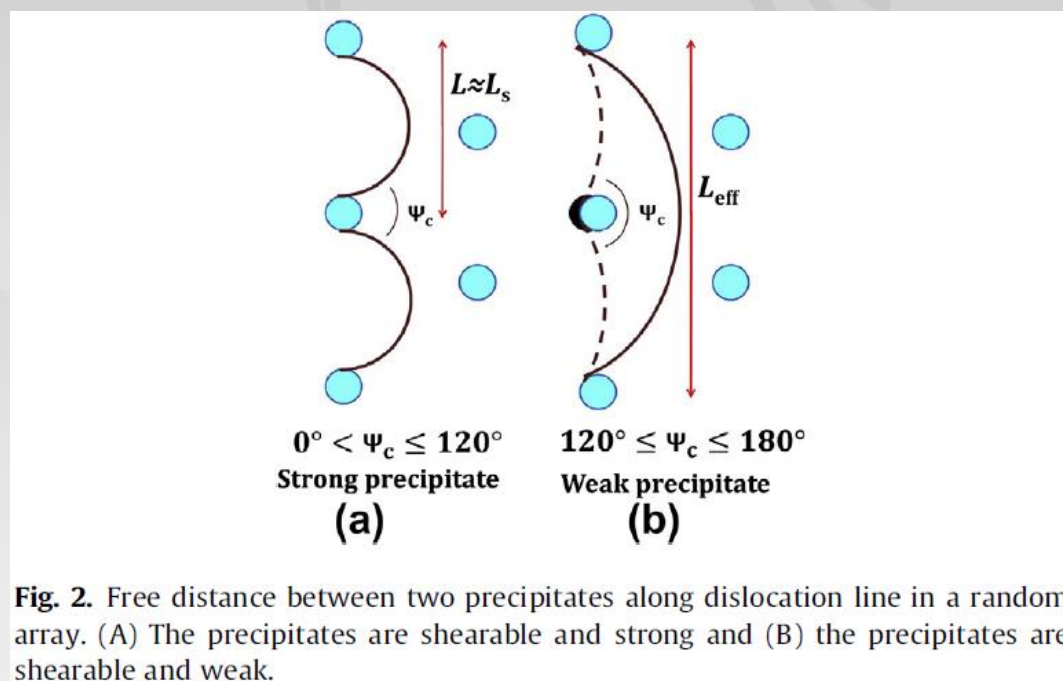
$$L_S = \sqrt{\frac{\ln 3}{2\pi \sum_{class} N_{V,class} r_{m,class}} + 4r_{ss}^2} - 2r_{ss}$$

Strong  
particles

$$r_{ss} = \sqrt{\frac{2}{3} \frac{\sum_{class} N_{V,class} r_{m,class}^2}{\sum_{class} N_{V,class} r_{m,class}}}$$

# „Weak“ vs. „strong“ particles

- Criterion: Dislocation bending angle  $\psi$  threshold
  - Strong resistance of particles  $\rightarrow$  High curvature of dislocation line  $\rightarrow$  small  $\psi$



Strong particles

$$L_S = \sqrt{\frac{\ln 3}{2\pi \sum_{class} N_{V,class} r_{m,class}} + 4r_{ss}^2} - 2r_{ss}$$

Weak particles

$$L_{eff} = L_S \left[ \cos\left(\frac{\psi}{2}\right) \right]^{-1/2}$$

# „Weak“ vs. „strong“ particles

- Dislocation line tension,  $T$ 
  - Simple model

$$T = \frac{Gb^2}{2}$$

$G$  - Shear modulus

$b$  - Burgers vector

$\nu$  - Poisson's ratio

$r_i$  - dislocation core radius

- Advanced model (different values for „weak“ and „strong“ particles)

$$T_{strong} = \frac{Gb^2}{4\pi} \left( \frac{1 + \nu - 3\nu \sin^2 \theta}{1 - \nu} \right) \ln \left( \frac{L_s}{r_i} \right)$$

$$T_{weak} = \frac{Gb^2}{4\pi} \left( \frac{1 + \nu - 3\nu \sin^2 \theta}{1 - \nu} \right) \ln \left( \frac{L_{eff}}{r_i} \right)$$

# „Weak“ vs. „strong“ particles

- Dislocation line tension,  $T$ 
  - Simple model

$$T = \frac{Gb^2}{2}$$

- Advanced model (different values for „weak“ and „strong“ particles)

$$T_{strong} = \frac{Gb^2}{4\pi} \left( \frac{1 + \nu - 3\nu \sin^2 \theta}{1 - \nu} \right) \ln \left( \frac{L_s}{r_i} \right)$$

$$T_{weak} = \frac{Gb^2}{4\pi} \left( \frac{1 + \nu - 3\nu \sin^2 \theta}{1 - \nu} \right) \ln \left( \frac{L_{eff}}{r_i} \right)$$

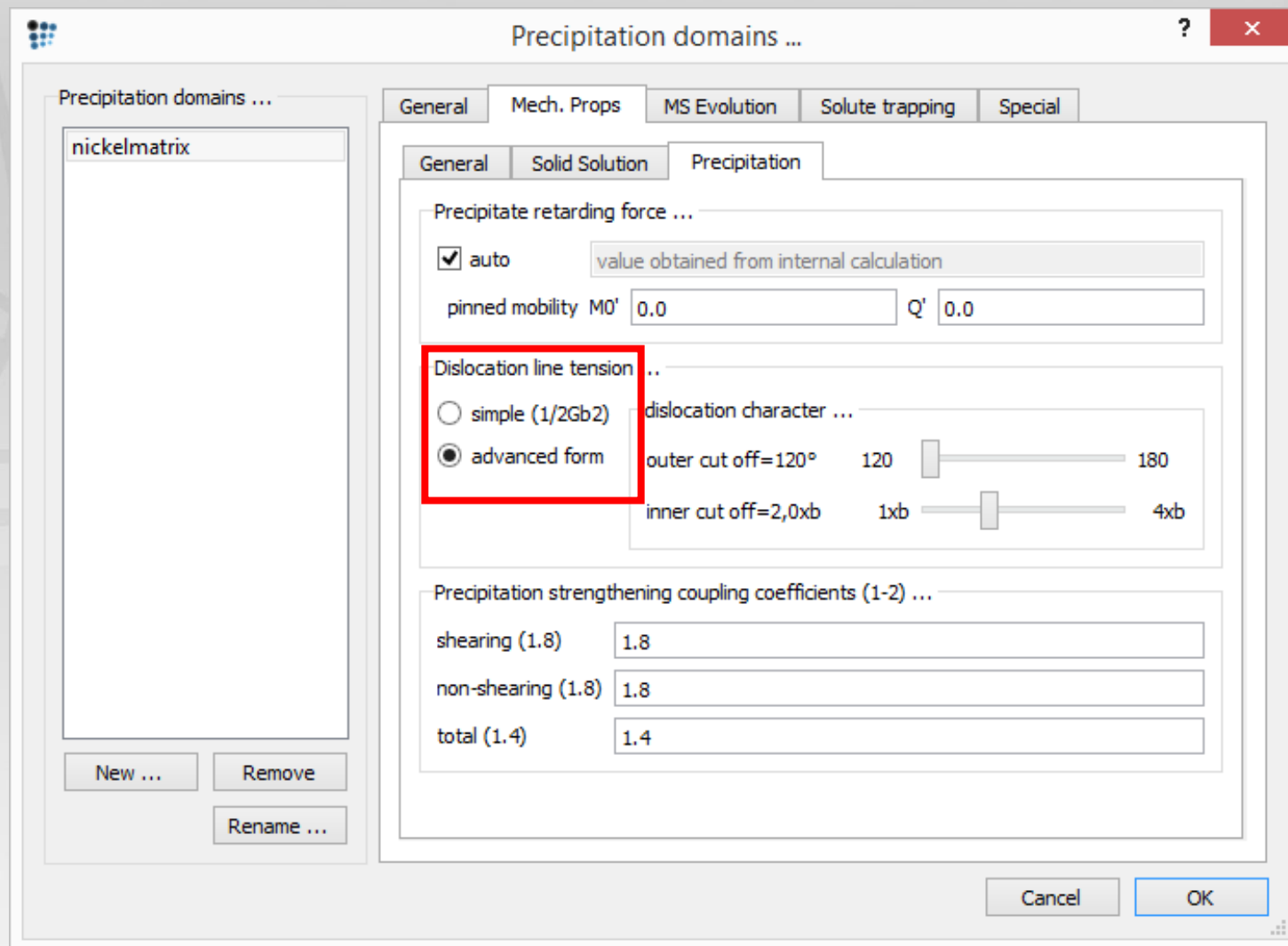
variables	value
kinetics: prec. strength	
DLT_SIMPLE\$*	
DLT_SIMPLE\$GAMMA_PRIME_P0	1.92907e-09
DLT_WEAK\$GAMMA_PRIME_P0	1.6019e-09
DLT_STRONG\$GAMMA_PRIME_P0	1.46929e-09

category: kinetics: prec. strength  
expression: DLT\_SIMPLE\$\*  
legal unit qualifiers: \*none\*  
-> dislocation line tension from simple description (1/2Gb^2)

# „Weak“ vs. „strong“ particles

- Dislocation line tension,  $T$

- Simple
- Advanced



# Shearable particles

- Models for “weak” and “strong” particles
- Various effects (contributions) considered
  - Coherency effect
  - Modulus effect
  - Anti-phase boundary effect
  - Stacking fault effect
  - Interfacial effect

# Coherency effect

- Strain field due to precipitation/matrix misfit
  - Strong particles

$$\tau_{coh,strong} = \frac{(2 \cos^2 \theta + 2.1352 \sin^2 \theta)}{L_s} \left( \frac{T_{strong}^3 \overset{\downarrow}{G \varepsilon r_m}}{b^3} \right)^{1/4}$$

$$\varepsilon = \frac{2}{3} \Delta_{lin} = \frac{2}{9} \Delta_{vol}$$

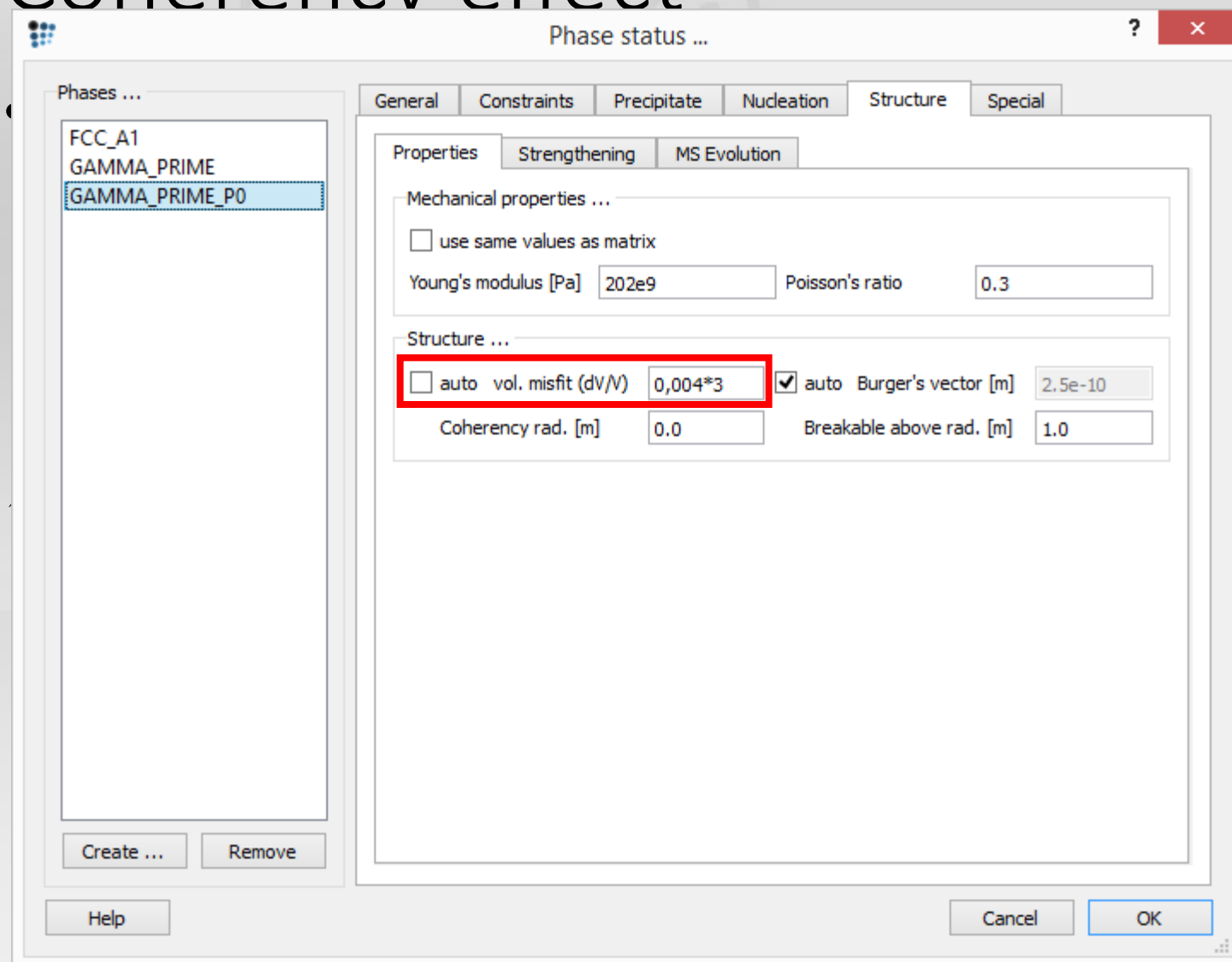
- Weak particles

$$\tau_{coh,weak} = \frac{(1.3416 \cos^2 \theta + 4.1127 \sin^2 \theta)}{L_s} \left( \frac{\overset{\downarrow}{G^3 \varepsilon^3 r_{eq}^3 b}}{T_{weak}} \right)^{1/2}$$

$\Delta_{lin}$  - Linear misfit

$\Delta_{vol}$  - Volumetric misfit

# Coherency effect

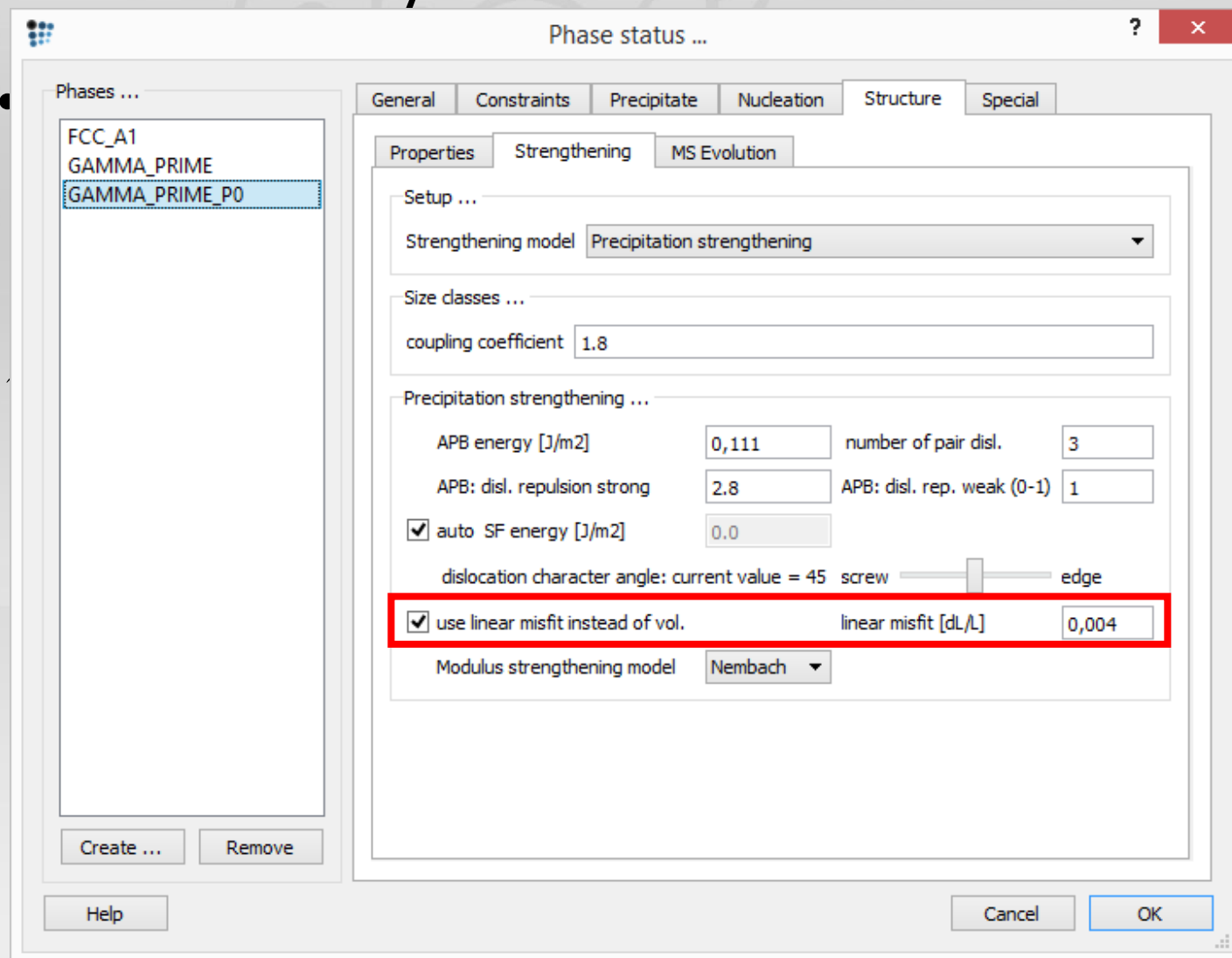


$$\varepsilon = \frac{2}{3} \Delta_{lin} = \frac{2}{9} \Delta_{vol}$$

$\Delta_{lin}$  - Linear misfit

$\Delta_{vol}$  - Volumetric misfit

# Coherency effect



$$\varepsilon = \frac{2}{3} \Delta_{lin} = \frac{2}{9} \Delta_{vol}$$

$\Delta_{lin}$  - Linear misfit

$\Delta_{vol}$  - Volumetric misfit

# Coherency effect

- Strain field due to precipitation/matrix misfit
  - Strong particles

$$\tau_{coh,strong} = \frac{(2 \cos^2 \theta + 2.1352 \sin^2 \theta)}{L_S} \left( \frac{T_{strong}^3 G \epsilon r_m}{b^3} \right)^{1/4}$$

- Weak particles

$$\tau_{coh,weak} = \frac{(1.3416 \cos^2 \theta + 4.1127 \sin^2 \theta)}{L_S} \left( \frac{G^3 \epsilon^3 r_{eq}^3 b}{T_{weak}} \right)^{1/2}$$

variables	value
▾ kinetics: prec. strength	
▾ TAO_COHER_WEAKS*	
TAO_COHER_WEAKSGAMMA_PRIME_P0	8.9264e+07
▾ TAO_COHER_STRONGS*	
TAO_COHER_STRONGSGAMMA_PRIME_P0	4.92374e+08

category: kinetics: prec. strength  
 expression: TAO\_COHER\_WEAK\$GAMMA\_PRIME\_P0  
 legal unit qualifiers: \*none\*  
 -> coherency hardening shear stress for shearable weak precipitates of individual phase

# Shearable particles

- Models for “weak” and “strong” particles
- Various effects (contributions) considered
  - Coherency effect
  - Modulus effect
  - Anti-phase boundary effect
  - Stacking fault effect
  - Interfacial effect

# Modulus effect

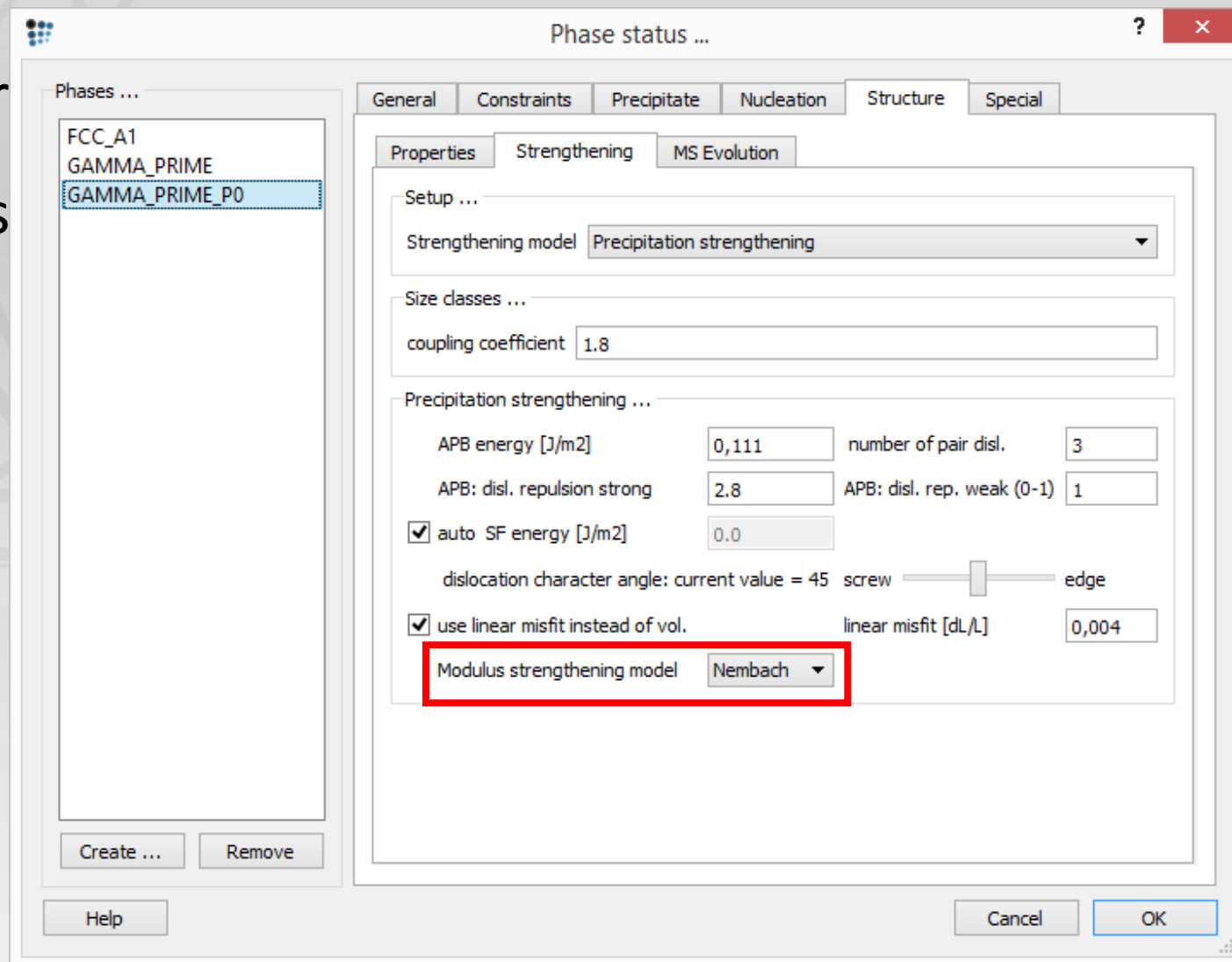
- Elastic properties of precipitate and matrix differ → dislocation energy inside and outside the precipitate differ
- 2 models
  - Nembach
  - Siems

# Modulus effect

- Elastic properties of precipitates and matrix  
energy inside and outside

- 2 models

- Nembach
- Siems



# Modulus effect

- Nembach model
  - Strong particles

$$\tau_{\text{mod,strong}} = \frac{F_{\text{mod}}}{bL_S}$$

$$F_{\text{mod}} = 0.05 |G - G_P| b^2 \left( \frac{r_{eq}}{b} \right)^{0.85}$$

- Weak particles

$$\tau_{\text{mod,weak}} = \frac{2T_{\text{weak}}}{bL_S} \left( \frac{F_{\text{mod}}}{2T_{\text{weak}}} \right)^{3/2}$$

$G_P$  - Particle shear modulus

# Modulus effect

- Siems model

$\nu_p$  - Particle Poisson ratio

Strong

$$\tau_{\text{mod,strong}} = 0.8 \frac{2T_{\text{strong}}}{bL_S} \left[ 1 - \left( \frac{E_p}{E} \right)^2 \right]^{1/2}$$

$$\frac{E_p}{E} = \frac{G_P(1-\nu) \log \frac{r_{eq}}{r_i} + G(1-\nu_p) \log \frac{L_S}{r_{eq}}}{G(1-\nu_p) \log \frac{L_S}{r_i}}$$

Weak

$$\tau_{\text{mod,weak}} = \frac{2T_{\text{weak}}}{bL_S} \left[ 1 - \left( \frac{E_p}{E} \right)^2 \right]^{3/4}$$

$$\frac{E_p}{E} = \frac{G_P(1-\nu) \log \frac{r_{eq}}{r_i} + G(1-\nu_p) \log \frac{L_{eff}}{r_{eq}}}{G(1-\nu_p) \log \frac{L_{eff}}{r_i}}$$

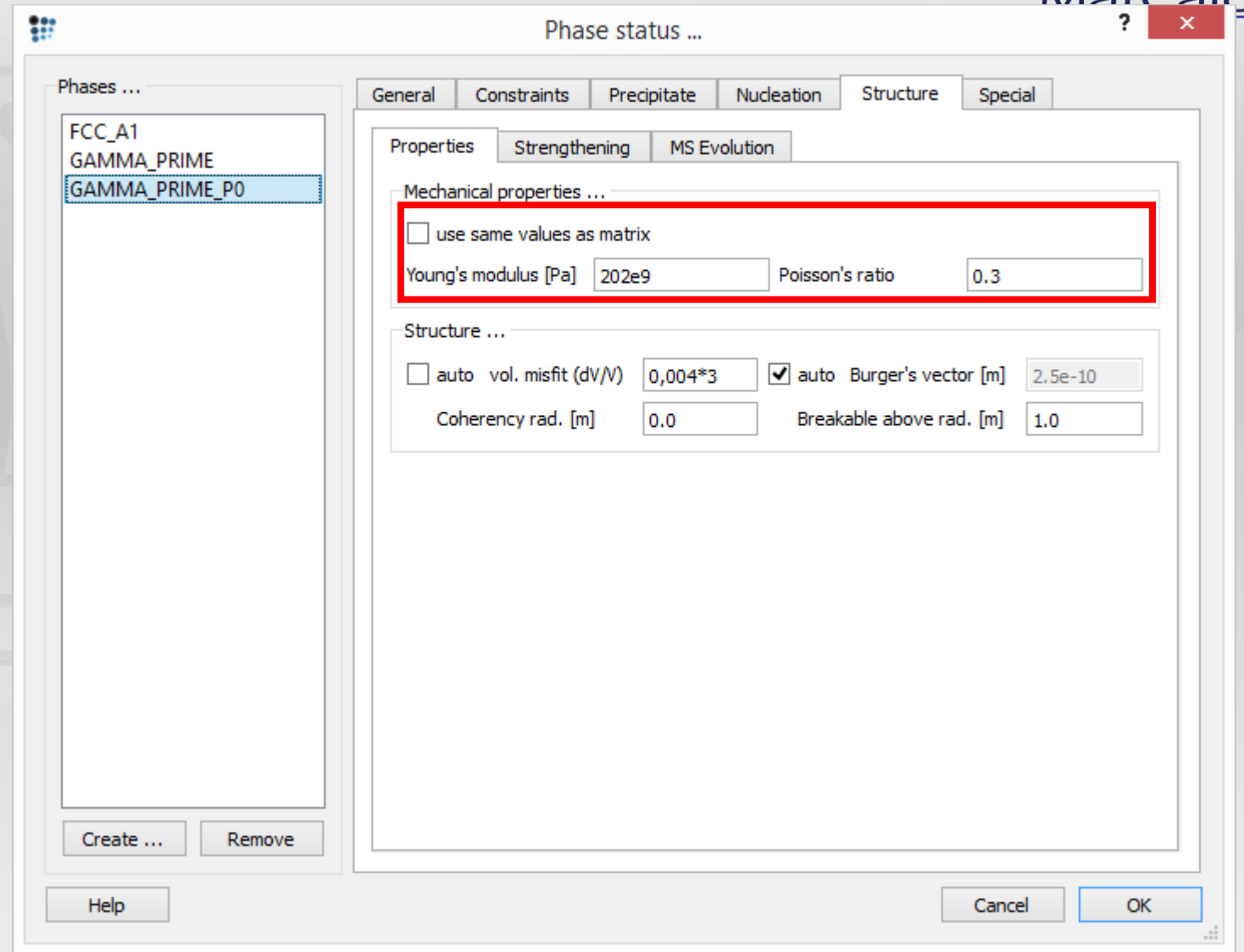
# Modulus effect

- Nembach model
  - Strong particles

$$\tau_{\text{mod,strong}} = \frac{F_{\text{mod}}}{bL_S}$$

- Weak particles

$$\tau_{\text{mod,weak}} = \frac{2T_{\text{weak}}}{bL_S} \left( \frac{F_{\text{mod}}}{2T_{\text{weak}}} \right)^{3/2}$$



$\nu_p$  - Particle Poisson ratio       $G_p$  - Particle shear modulus

# Modulus effect

- Elastic properties of precipitate and matrix differ → dislocation energy inside and outside the precipitate differ
- 2 models
  - Nembach
  - Siems

$$\tau_{\text{mod,weak}}$$

$$\tau_{\text{mod,strong}}$$

variables	value
kinetics: prec. strength	
TAO_MOD_WEAKS*	
TAO_MOD_WEAK\$GAMMA_PRIME_P0	1.80848e+06
TAO_MOD_STRONGS*	
TAO_MOD_STRONG\$GAMMA_PRIME_P0	1.32552e+07

category: kinetics: prec. strength  
 expression: TAO\_MOD\_WEAK\$GAMMA\_PRIME\_P0  
 legal unit qualifiers: \*none\*  
 -> modulus mismatch hardening shear stress for weak shearable precipitates of individual phase

# Shearable particles

- Models for “weak” and “strong” particles
- Various effects (contributions) considered
  - Coherency effect
  - Modulus effect
  - Anti-phase boundary effect
  - Stacking fault effect
  - Interfacial effect

# Anti-phase boundary (APB) effect

- Dislocation passing through ordered precipitate increases the energy by creating the APB

- Strong particles

$$\tau_{APB, strong} = \frac{0.69}{bL_S} \left( \frac{8wT_{strong}r_{eq}\gamma_{APB}}{3} \right)^{1/2}$$

$\gamma_{APB}$  - APB energy

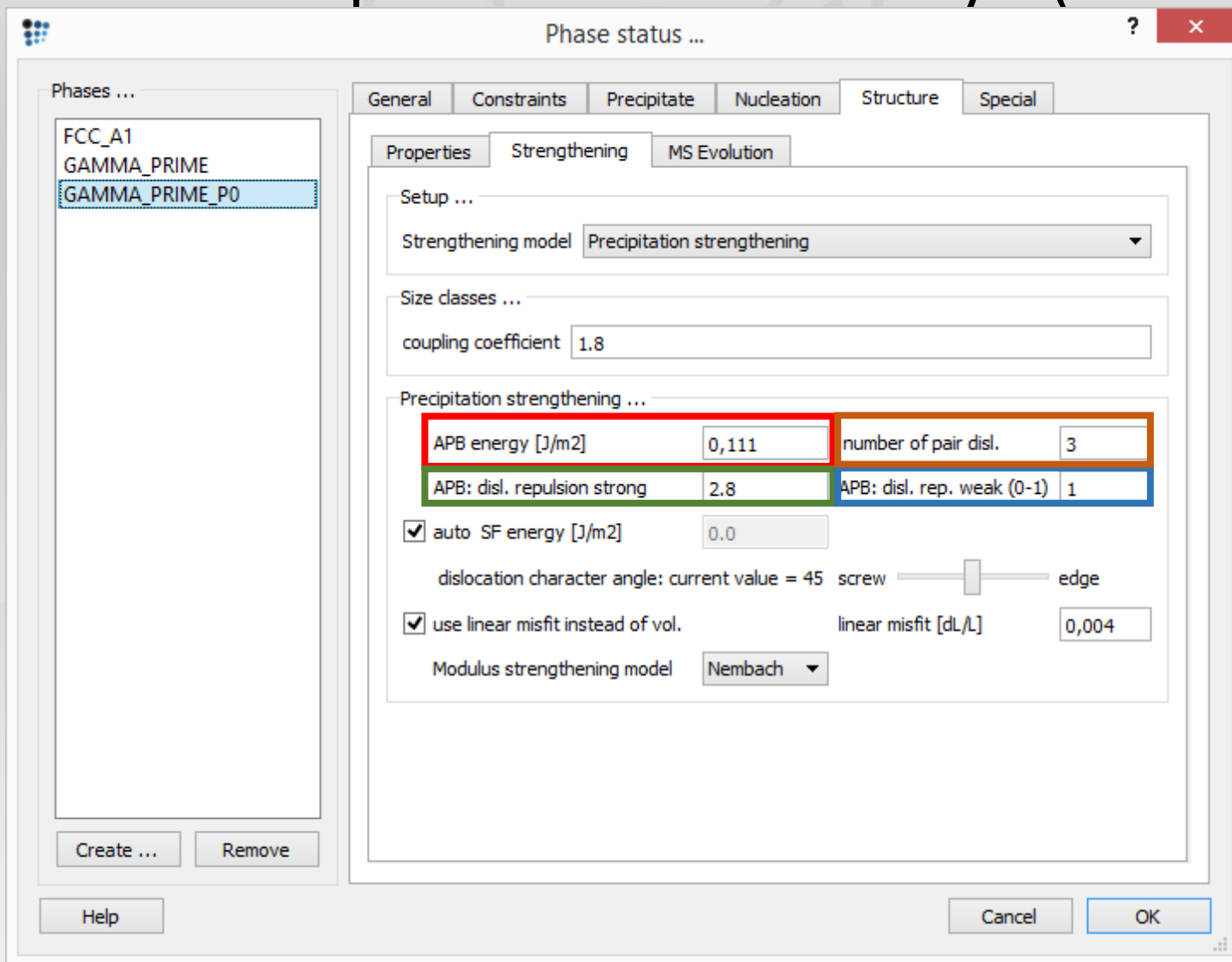
$w, \beta$  - Interaction parameter between the leading and trailing dislocation

$s$  - Number of pair dislocations

- Weak particles

$$\tau_{APB, weak} = \frac{2}{sbL_S} \left[ 2T_{weak} \left( \frac{r_{eq}\gamma_{APB}}{T_{weak}} \right)^{3/2} - \frac{16\beta\gamma_{APB}r_{eq}^2}{3\pi L_S} \right]$$

# Anti-phase boundary (APB) effect



precipitate increases the energy

$\gamma_{APB}$  - APB energy

$w$   $\beta$  - Interaction parameter between

the leading and trailing dislocation

$s$  - Number of pair dislocations

# Anti-phase boundary (APB) effect

- Dislocation passing through ordered precipitate increases the energy by creating the APB
  - Strong particles

$$\tau_{APB, strong} = \frac{0.69}{bL_S} \left( \frac{8wT_{strong}r_{eq}\gamma_{APB}}{3} \right)^{1/2}$$

- Weak particles

$$\tau_{APB, weak} = \frac{2}{sbL_S} \left[ 2T_{weak} \left( \frac{r_{eq}\gamma_{APB}}{T_{weak}} \right)^{3/2} - \frac{16\beta\gamma_{APB}r_{eq}^2}{3\pi L_S} \right]$$

variables	value
kinetics: prec. strength	
TAO_APB_WEAK\$*	
TAO_APB_WEAK\$GAMMA_PRIME_P0	2.2558e+08
TAO_APB_STRONG\$*	
TAO_APB_STRONG\$GAMMA_PRIME_P0	7.78014e+08

category: kinetics: prec. strength  
 expression: TAO\_APB\_WEAK\$GAMMA\_PRIME\_P0  
 legal unit qualifiers: \*none\*  
 -> anti-phase boundary hardening shear stress for weak shearable precipitates of individual phase

# Shearable particles

- Models for “weak” and “strong” particles
- Various effects (contributions) considered
  - Coherency effect
  - Modulus effect
  - Anti-phase boundary effect
  - Stacking fault effect
  - Interfacial effect

# Stacking fault (SF) effect

- Passing dislocation creates a stacking fault – energy difference between the SF in the precipitate and matrix

$$K_{SF} = \frac{Gb_p^2(2 - \nu - 2\nu \cos(2\theta))}{8\pi(1 - \nu)}$$

$$W_{eff} = \frac{2K_{SF}}{\gamma_{SFM} + \gamma_{SFP}}$$

$$F_{SF} = 2(\gamma_{SFM} - \gamma_{SFP})\sqrt{W_{eff}r_{eq} - W_{eff}^2/4}$$

$b_p$  - Burger's vector of particle

$\gamma_{SFP}$  - Stacking fault energy of particle

$\gamma_{SFM}$  - Stacking fault energy of matrix



# Stacking fault (SF) effect

- Passing dislocation creates a stacking fault – energy difference between the SF in the precipitate and matrix
- Strong particles

$$\tau_{SF, strong} = \frac{F_{SF}}{bL_S}$$

$$K_{SF} = \frac{Gb_p^2(2 - \nu - 2\nu \cos(2\theta))}{8\pi(1 - \nu)}$$

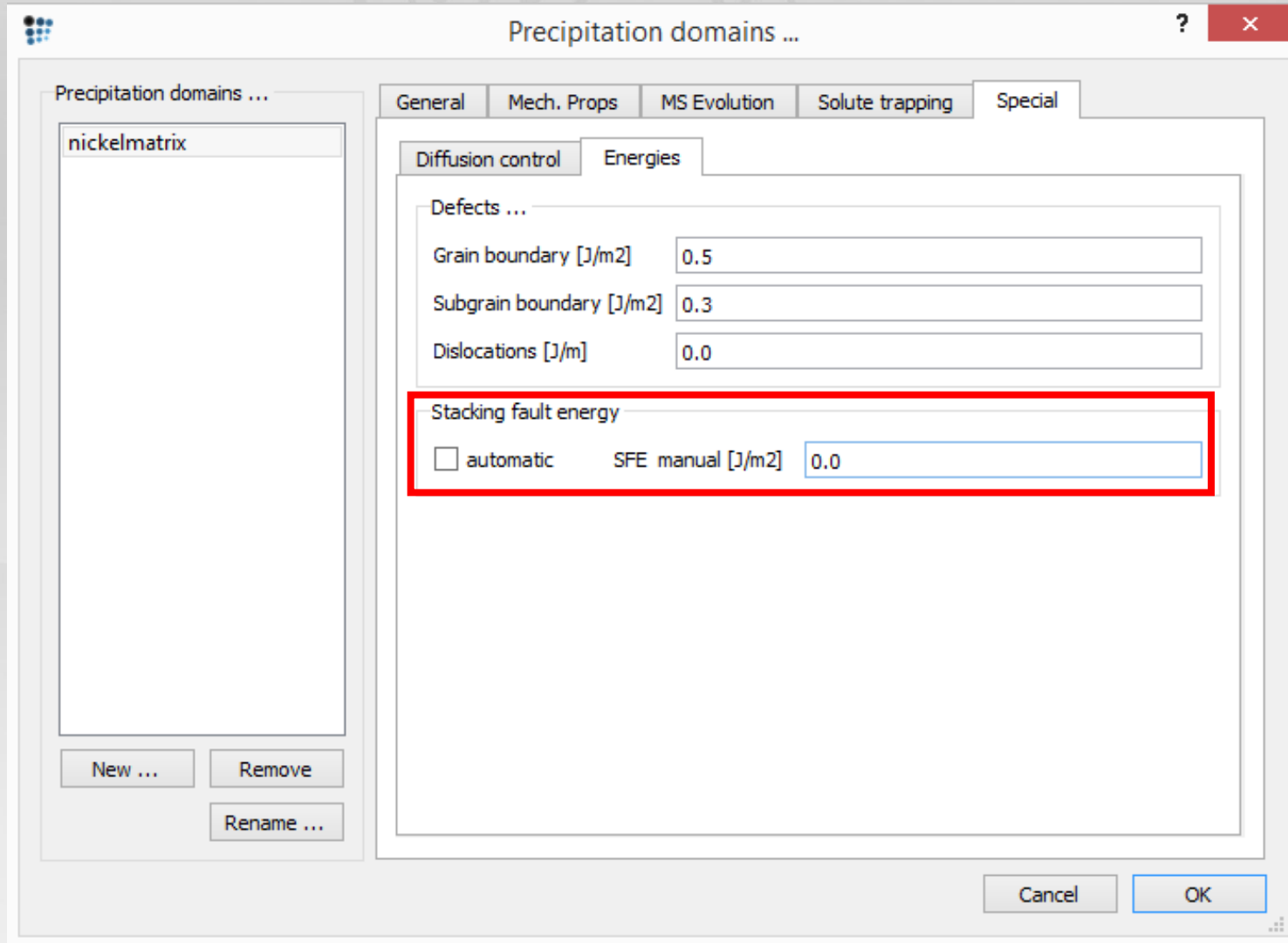
- Weak particles

$$\tau_{SF, weak} = \frac{2T_{weak}}{bL_S} \left( \frac{F_{SF}}{2T_{weak}} \right)^{3/2}$$

$$W_{eff} = \frac{2K_{SF}}{\gamma_{SFM} + \gamma_{SFP}}$$

$$F_{SF} = 2(\gamma_{SFM} - \gamma_{SFP}) \sqrt{W_{eff} r_{eq} - W_{eff}^2 / 4}$$

# Stacking fault (SF) effect



– energy difference

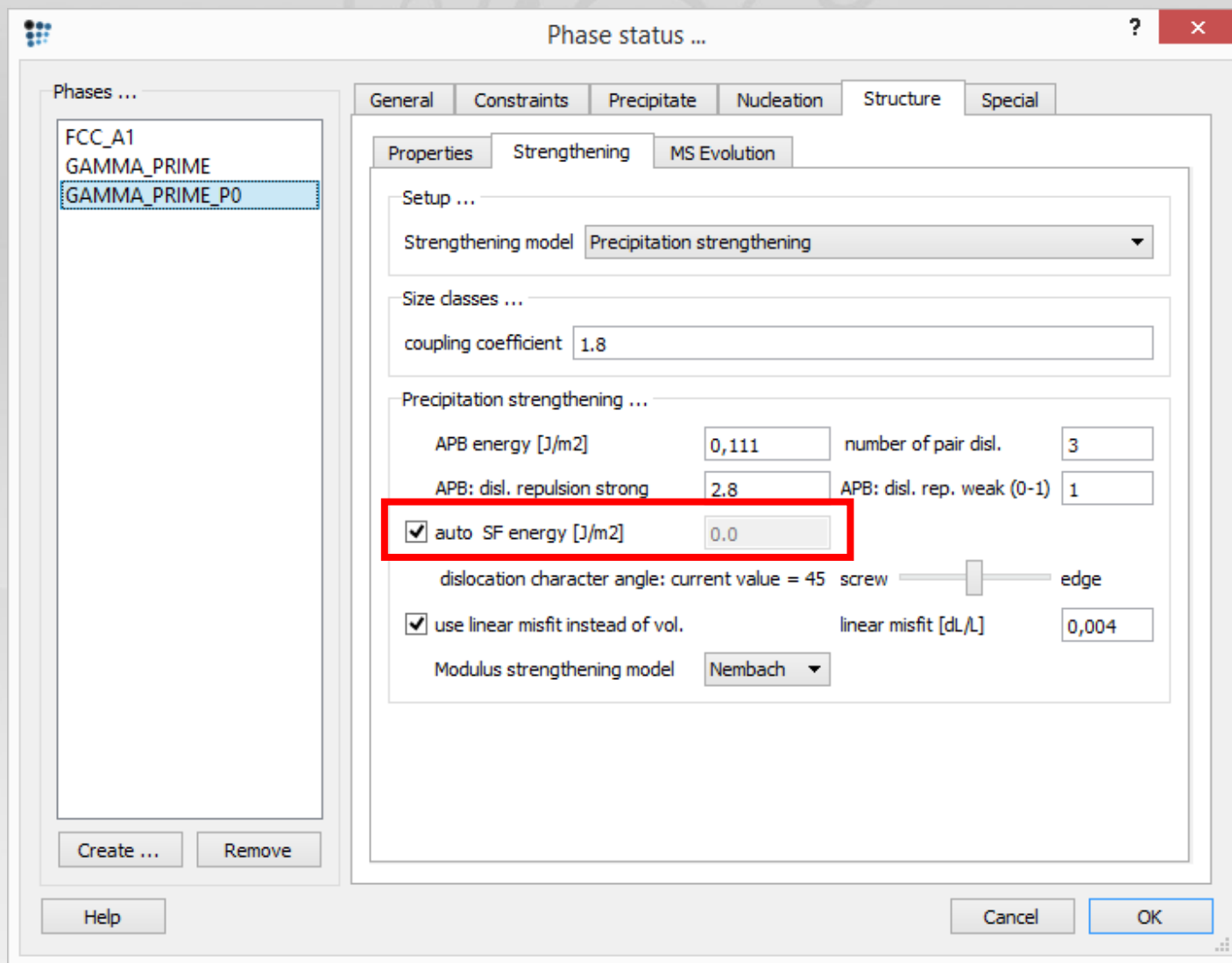
rix

$b_p$  - Burger's vector of particle

$\gamma_{SFP}$  - Stacking fault energy of particle

$\gamma_{SFM}$  - Stacking fault energy of matrix

# Stacking fault (SF) effect



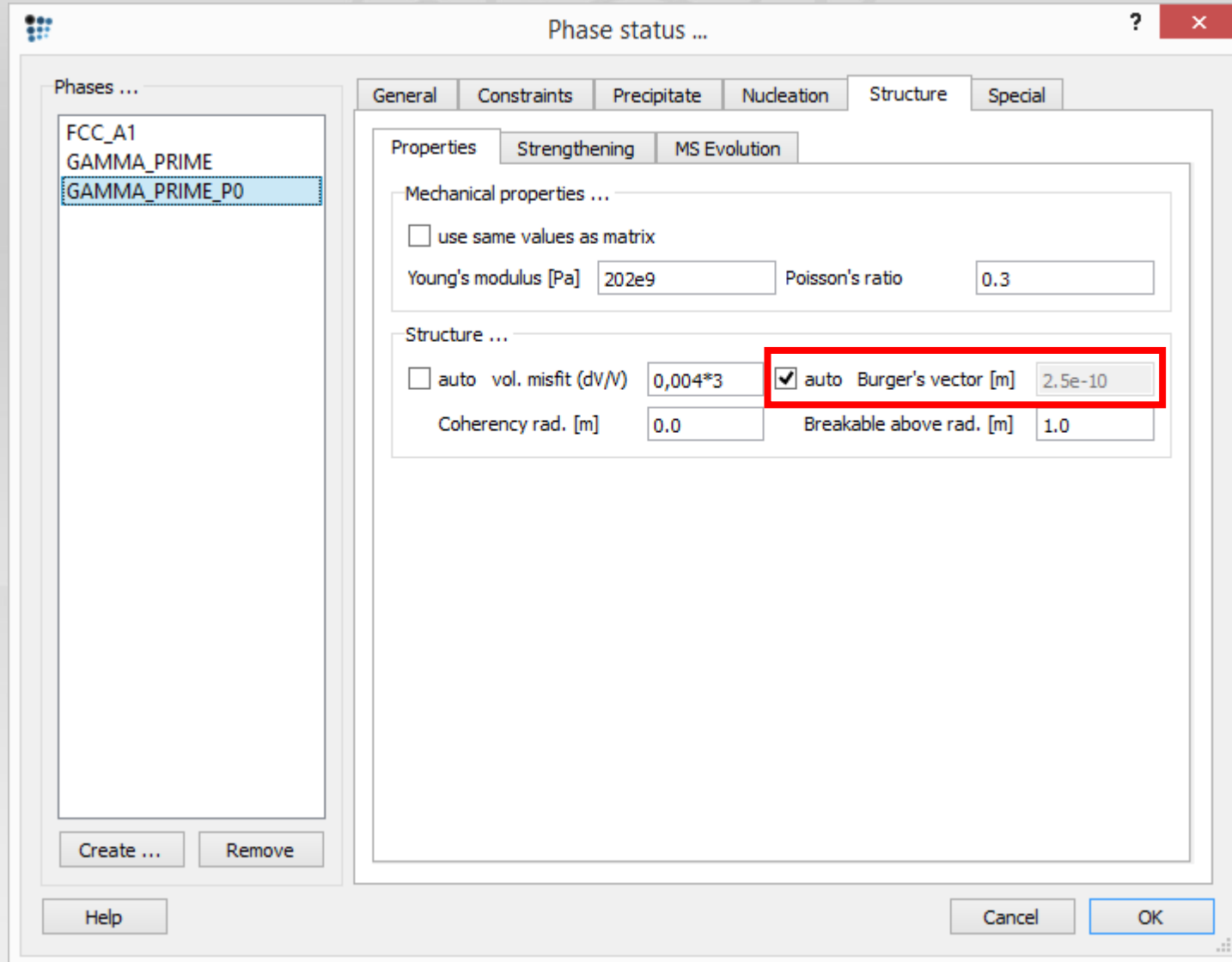
It – energy difference  
matrix

$b_p$  - Burger's vector of particle

$\gamma_{SFP}$  - Stacking fault energy of particle

$\gamma_{SFM}$  - Stacking fault energy of matrix

# Stacking fault (SF) effect



It – energy difference  
matrix

$b_p$  - Burger's vector of particle

$\gamma_{SFP}$  - Stacking fault energy of particle

$\gamma_{SFM}$  - Stacking fault energy of matrix

# Stacking fault (SF) effect

- Passing dislocation creates a stacking fault – energy difference between the SF in the precipitate and matrix
- Strong particles

$$\tau_{SF, strong} = \frac{F_{SF}}{bL_S}$$

- Weak particles

$$\tau_{SF, weak} = \frac{2T_{weak}}{bL_S} \left( \frac{F_{SF}}{2T_{weak}} \right)^{3/2}$$

variables	value
kinetics: prec. strength	
TAO_SFE_WEAKS*	
TAO_SFE_WEAK\$GAMMA_PRIME_P0	0
TAO_SFE_STRONGS*	
TAO_SFE_STRONG\$GAMMA_PRIME_P0	0

category: kinetics: prec. strength  
 expression: TAO\_SFE\_WEAK\$GAMMA\_PRIME\_P0  
 legal unit qualifiers: \*none\*  
 -> stacking fault energy hardening shear stress for weak shearable precipitates of individual phase

# Shearable particles

- Models for “weak” and “strong” particles
- Various effects (contributions) considered
  - Coherency effect
  - Modulus effect
  - Anti-phase boundary effect
  - Stacking fault effect
  - Interfacial effect

# Interfacial effect

- Passing dislocation increases the area of precipitate/matrix interface
  - Strong particles

$$\tau_{\text{int, strong}} = \frac{F_{\text{int}}}{bL_S}$$

$$F_{\text{int}} = 2\gamma_{PM}b$$

- Weak particles

$$\tau_{\text{int, weak}} = \frac{2T_{\text{weak}}}{bL_S} \left( \frac{F_{\text{int}}}{2T_{\text{weak}}} \right)^{3/2}$$

$\gamma_{PM}$  - Precipitate/matrix interface energy

# Interfacial effect

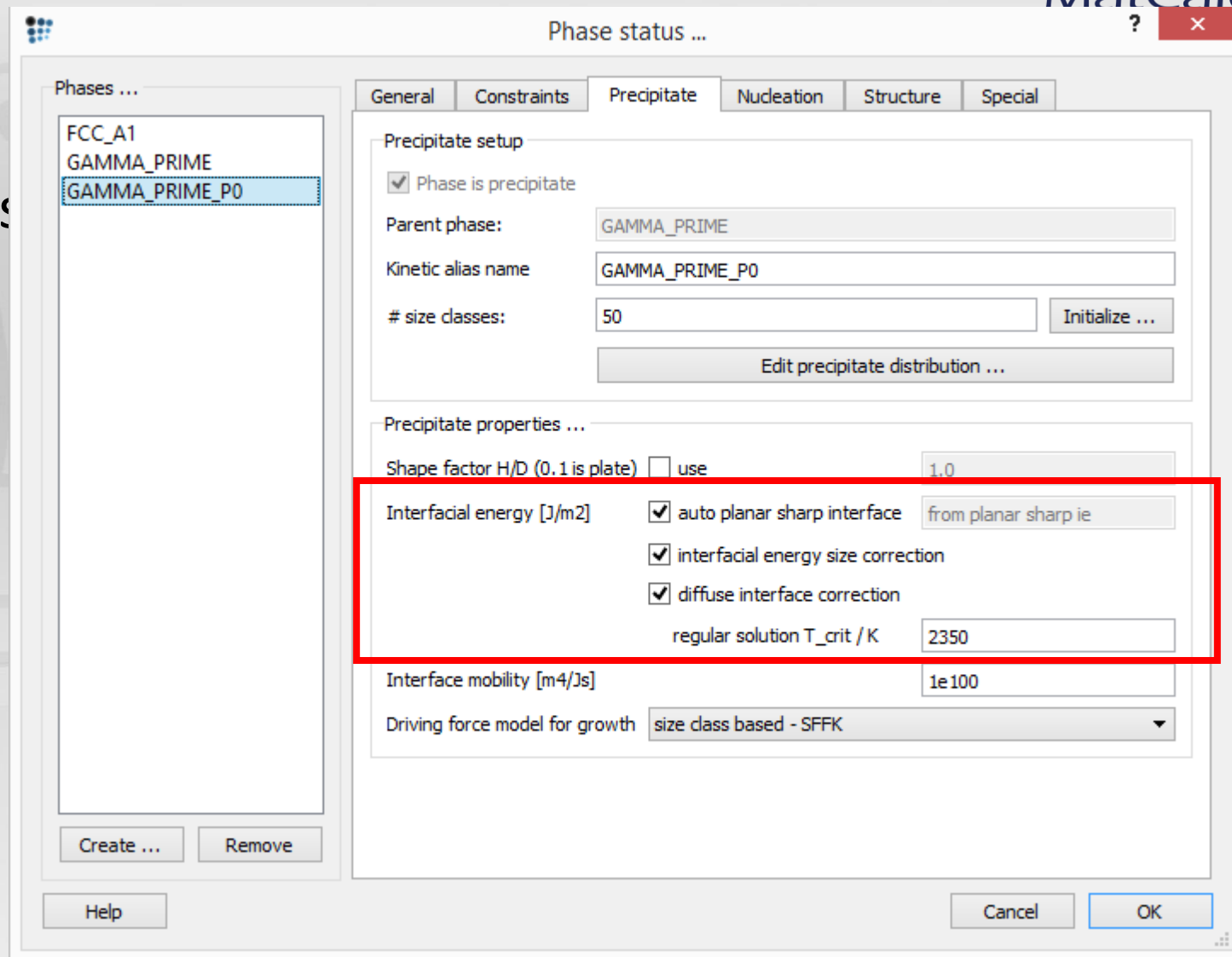
- Passing dislocation increases

- Strong particles

$$\tau_{int, strong} = \frac{F_{int}}{bL_S}$$

- Weak particles

$$\tau_{int, weak} = \frac{2T_{weak}}{bL_S} \left( \frac{F_{int}}{2T_{weak}} \right)^{3/2}$$



$\gamma_{PM}$  - Precipitate/matrix interface energy

# Interfacial effect

- Passing dislocation increases the area of precipitate/matrix interface
  - Strong particles

$$\tau_{\text{int, strong}} = \frac{F_{\text{int}}}{bL_S}$$

- Weak particles

$$\tau_{\text{int, weak}} = \frac{2T_{\text{weak}}}{bL_S} \left( \frac{F_{\text{int}}}{2T_{\text{weak}}} \right)^{3/2}$$

variables	value
kinetics: prec. strength	
TAO_CHEM_WEAKS*	
TAO_CHEM_WEAK\$GAMMA_PRIME_P0	2.30719e+06
TAO_CHEM_STRONGS*	
TAO_CHEM_STRONG\$GAMMA_PRIME_P0	1.55919e+07

category: kinetics: prec. strength  
 expression: TAO\_CHEM\_WEAK\$GAMMA\_PRIME\_P0  
 legal unit qualifiers: \*none\*  
 -> chemical hardening shear stress for shearable weak precipitates of individual phase

# Size distribution dependent strengthening

- Precipitate size dependence
- Some general parameters/settings
- 2 scenarios for dislocation behavior:
  - Non-shearable particles (Orowan mechanism) → bypassing precipitate
  - Shearable particles (weak or strong) → cutting precipitate
- Critical stresses for both scenarios are evaluated

# Identifying the strengthening regime

- Values of  $\tau$  evaluated for each of three regimes (Non-shearable, shearable weak, shearable strong)
  - Shearable model taken as a superposition of individual contributions

$$\tau_{i,strong} = \left( \tau_{i,coher,strong}^{m_{sh}} + \tau_{i,mod,strong}^{m_{sh}} + \tau_{i,APB,strong}^{m_{sh}} + \tau_{i,SF,strong}^{m_{sh}} + \tau_{i,int,strong}^{m_{sh}} \right)^{1/m_{sh}}$$

$$\tau_{i,weak} = \left( \tau_{i,coher,weak}^{m_{sh}} + \tau_{i,mod,weak}^{m_{sh}} + \tau_{i,APB,weak}^{m_{sh}} + \tau_{i,SF,weak}^{m_{sh}} + \tau_{i,int,weak}^{m_{sh}} \right)^{1/m_{sh}}$$

$$\tau_{i,nsh}$$

$$i - \begin{cases} \text{Precipitate phase (for „mean radius“ models)} \\ \text{Size class (for „multi-class“ model)} \end{cases}$$

# Identifying the strengthening regime

- Values of  $\tau$  evaluated for shearable weak, shearable strong
- Shearable model taken

$$\tau_{i, strong} = \left( \tau_{i, coher, strong}^{m_{sh}} + \tau_{i, nsh} \right)$$

$$\tau_{i, weak} = \left( \tau_{i, coher, weak}^{m_{sh}} + \tau_{i, nsh} \right)$$

$$\tau_{i, nsh}$$

Precipitation domains ...

nickelmatrix

General Mech. Props MS Evolution Solute trapping Special

General Solid Solution Precipitation

Precipitate retarding force ...

auto value obtained from internal calculation

pinned mobility M0' 0.0 Q' 0.0

Dislocation line tension ...

simple (1/2Gb2) dislocation character ...

advanced form

outer cut off=120° 120 180

inner cut off=2,0xb 1xb 4xb

Precipitation strengthening coupling coefficients (1-2) ...

shearing (1.8)	1.8
non-shearing (1.8)	1.8
total (1.4)	1.4

New ... Remove Rename ...

Cancel OK

# Identifying the strengthening regime

- Lowest value regime taken as the operative one
  - Precipitate „releases“ passing dislocation according to „lowest resistance“ scenario.

$$\tau_i = \min. (\tau_{i,nsh}, \tau_{i,strong}, \tau_{i,weak})$$

$$\tau_{i,strong} = \left( \tau_{i,coher,strong}^{m_{sh}} + \tau_{i,mod,strong}^{m_{sh}} + \tau_{i,APB,strong}^{m_{sh}} + \tau_{i,SF,strong}^{m_{sh}} + \tau_{i,int,strong}^{m_{sh}} \right)^{1/m_{sh}}$$

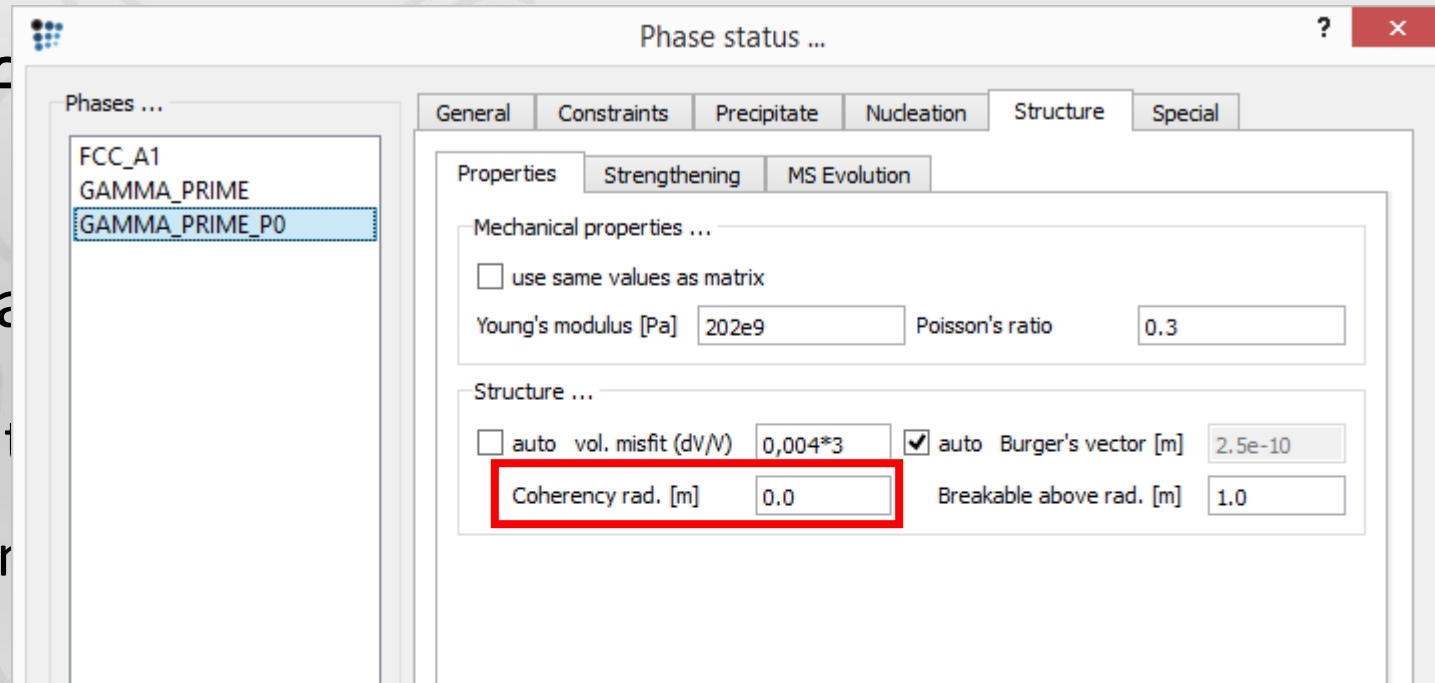
$$\tau_{i,weak} = \left( \tau_{i,coher,weak}^{m_{sh}} + \tau_{i,mod,weak}^{m_{sh}} + \tau_{i,APB,weak}^{m_{sh}} + \tau_{i,SF,weak}^{m_{sh}} + \tau_{i,int,weak}^{m_{sh}} \right)^{1/m_{sh}}$$

$$\tau_{i,nsh}$$

$$i - \begin{cases} \text{Precipitate phase (for „mean radius“ models)} \\ \text{Size class (for „multi-class“ model)} \end{cases}$$

# Identif

- Lowest va
- Precipit
- resistar



If coherency radius  $\neq 0$  and the respective radius (mean value or size class) is greater than the coherency radius  $\rightarrow$   
the non-shearable regime is taken for further calculation ( $\tau_i = \tau_{i,nsh}$ )

# Calculating strengthening contribution of individual precipitate phase, $\tau_j$

- “Multi-class” strengthening model selected
  - Superposition of individual size class contributions

$$\tau_j = \left[ \left( \sum_{i,sh} (\tau_{i,sh})^{m_{sh}} \right)^{m_{sum}/m_{sh}} + \left( \sum_{i,nsh} (\tau_{i,nsh})^{m_{nsh}} \right)^{m_{sum}/m_{nsh}} \right]^{1/m_{sum}}$$

$$\tau_{i,strong}, \tau_{i,weak} \rightarrow \tau_i \rightarrow \tau_{i,sh}$$

$$\tau_{i,nsh} \rightarrow \tau_i \rightarrow \tau_{i,nsh}$$

- Other strengthening models

$$\tau_j = \tau_i$$

# Calculating strenghtening contribution of individual precipitate phase, $\tau_j$

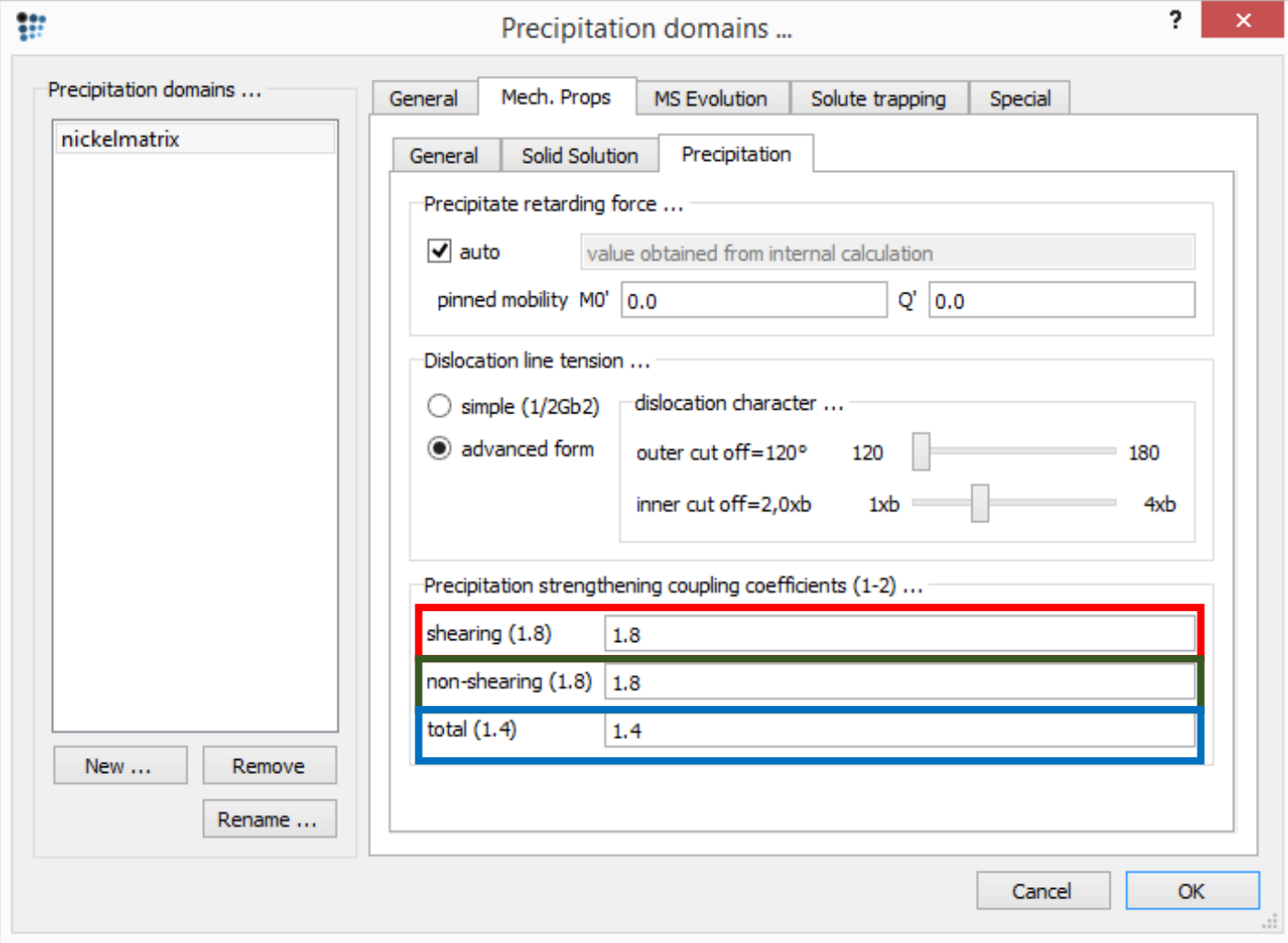
- “Multi-class” strengthening model
  - Superposition of individual sizes

$$\tau_j = \left[ \left( \sum_{i,sh} (\tau_{i,sh})^{m_{sh}} \right)^{\frac{m_{sum}}{m_{sh}}} + \left( \sum_{i,nsh} (\tau_{i,nsh})^{m_{nsh}} \right)^{\frac{m_{sum}}{m_{nsh}}} \right]^{\frac{m_{sum}}{m_{sh} + m_{nsh}}}$$

$\tau_{i,strong}, \tau_{i,weak} \rightarrow$   
 $\tau_{i,nsh} \rightarrow \tau_i \rightarrow \tau_j$

- Other strengthening models

$$\tau_j = \tau_i$$



# Total precipitation shear stress, $\tau_{prec}$

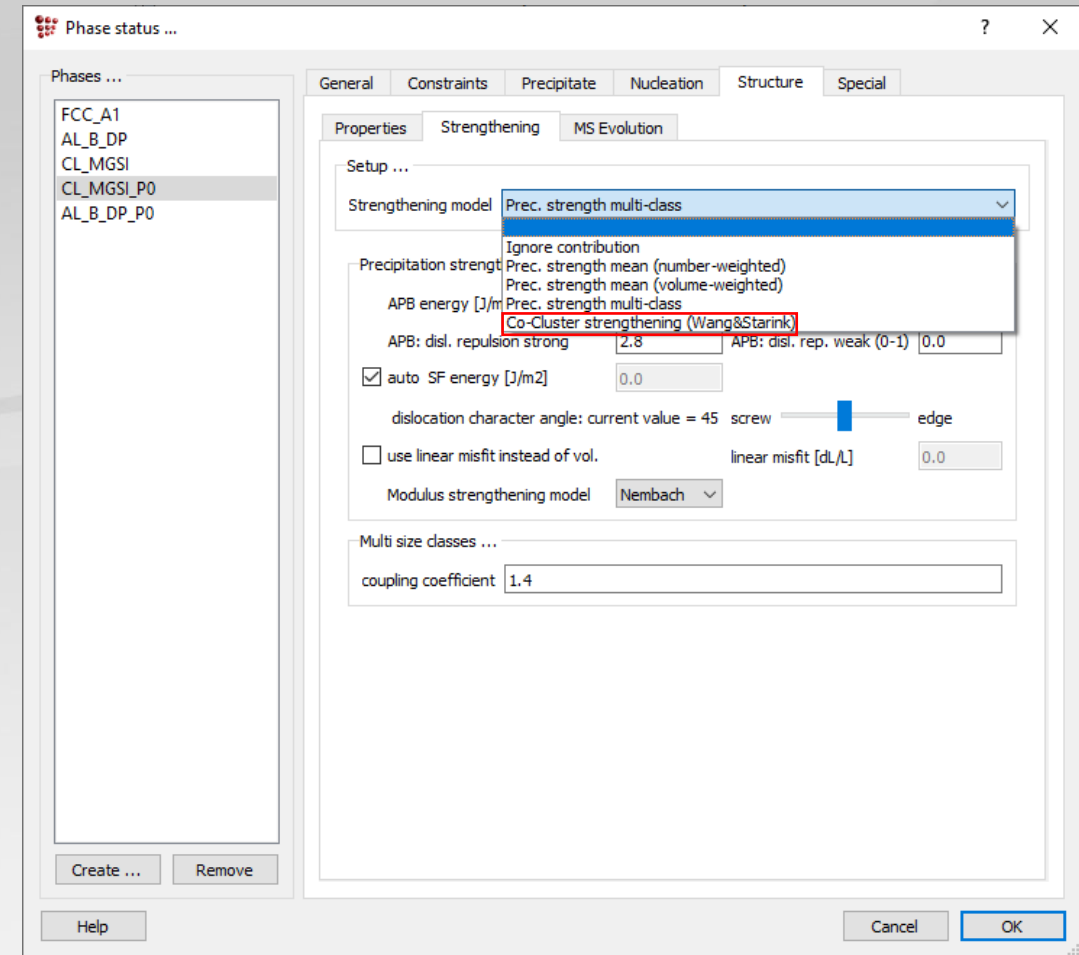
- Superposition of individual precipitate phase contributions

$$\tau_{prec} = \left[ \sum_j \tau_j^{9/5} \right]^{5/9}$$

- Precipitate phases with “grain boundaries” (including “edges” and “corners”) as nucleation sites are ignored ( $\tau_j = 0$ )

# Precipitation strengthening, $\sigma_{prec}$

- 2 alternative models available
  - Size distribution dependent strengthening
  - **Co-cluster strengthening**



# Co-cluster strengthening

- Related to binding enthalpy of co-cluster elements

(applicable only for phases containing two elements)

$$\tau_{prec} = \frac{8}{3\sqrt{3}} \frac{\Delta H_{ccl}}{N_A b^3} [f_{ccl} y_A (1 - x_B) + f_{ccl} y_B (1 - x_A) - 3x_A x_B]$$

$\Delta H_{ccl}$  - Binding enthalpy of heteroatomic bond in co-cluster

$b$  - Burgers vector

$N_A$  - Avogadro number

$y_i$  - Content of element  $i$  in co-cluster

$x_i$  - Content of element  $i$  in matrix

$f_{ccl}$  - Co-cluster phase fraction

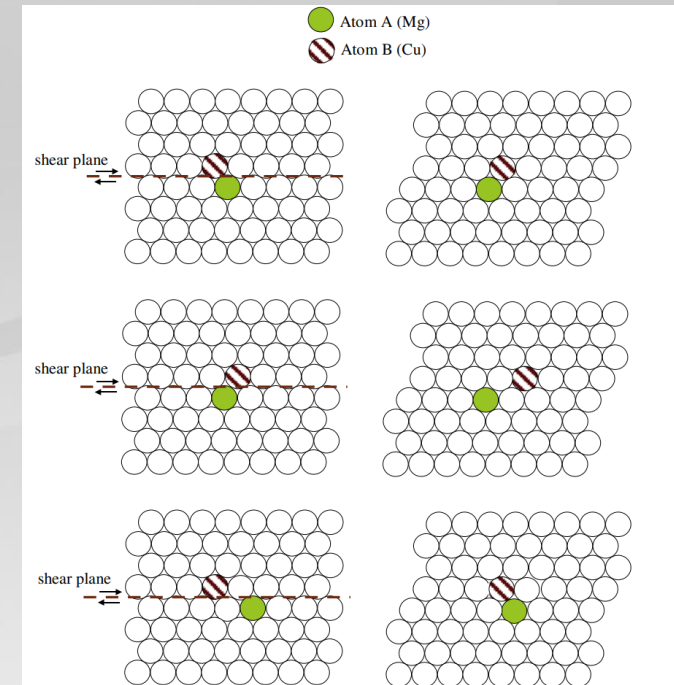


Fig. 2. A 111 plane in an FCC lattice with a 2-atom co-cluster being cut by a dislocation. Top shows before and after with the co-cluster remaining intact in a rotated form; middle shows before and after with the co-cluster being eliminated, which requires an energy input; and bottom shows before and after in the case where the passing of one dislocation creates a co-cluster, which releases energy.

# Co-cluster strengthening

- Related to binding enthalpy of co-

(applicable only for phases containing two ele

$$\tau_{prec} = \frac{8}{3\sqrt{3}} \frac{\Delta H_{ccl}}{N_A b^3} [f_{ccl} y_A (1 - x_B) + f_{ccl} y_B (1 - x_A)]$$

$\Delta H_{ccl}$  - Binding enthalpy of heteroatomic bond in co-cluster

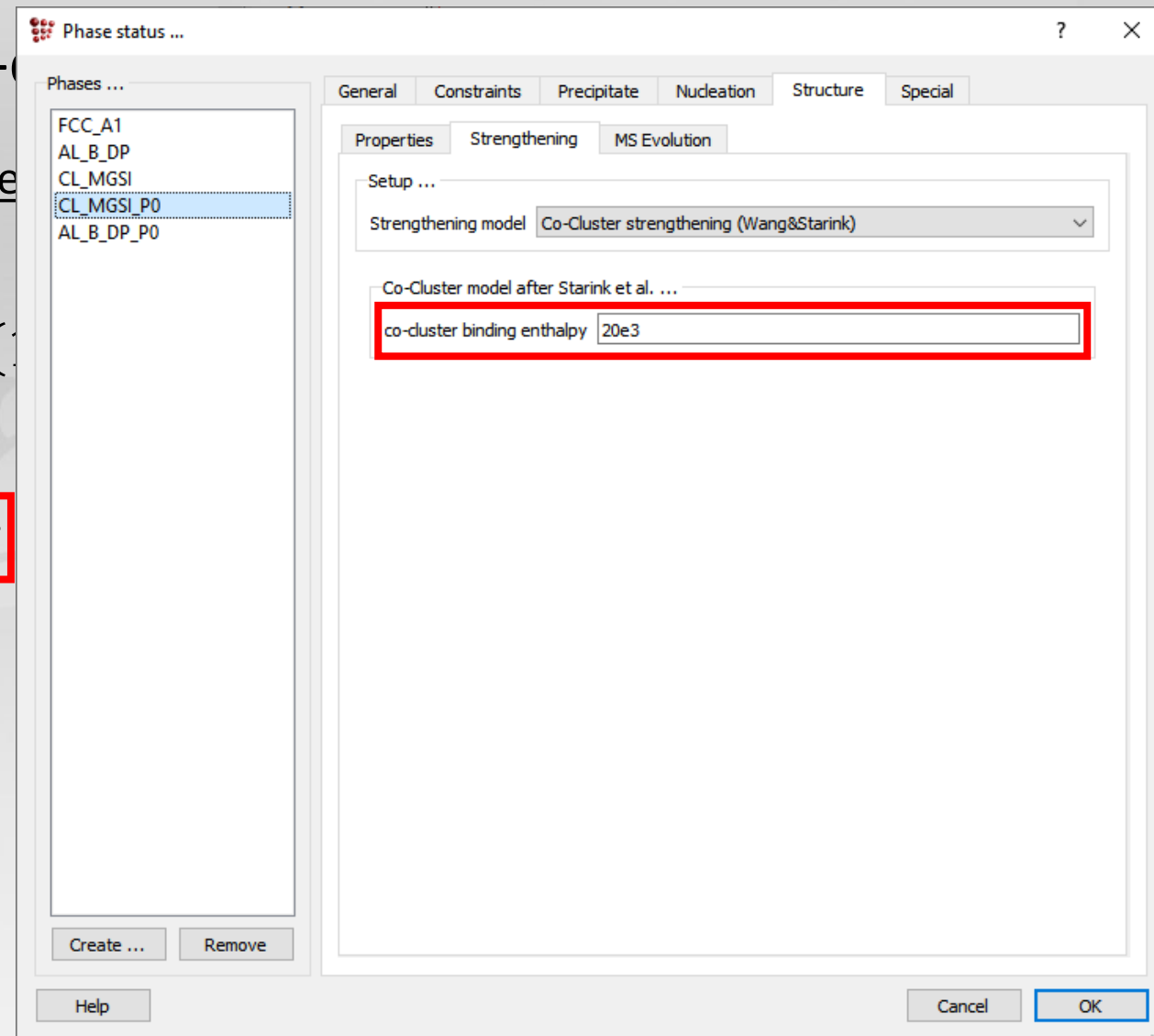
$b$  - Burgers vector

$N_A$  - Avogadro number

$y_i$  - Content of element  $i$  in co-cluster

$x_i$  - Content of element  $i$  in matrix

$f_{ccl}$  - Co-cluster phase fraction



# Co-cluster strengthening

- Related to binding enthalpy of co-cluster elements

(applicable only for phases containing two elements)

$$\tau_{prec} = \frac{8}{3\sqrt{3}} \frac{\Delta H_{ccl}}{N_A b^3} [f_{ccl} y_A (1 - x_B) + f_{ccl} y_B (1 - x_A) - 3x_A x_B]$$

$\Delta H_{ccl}$  - Binding enthalpy of heteroatomic bond in co-cluster

$b$  - Burgers vector

$N_A$  - Avogadro number

$y_i$  - Content of element  $i$  in co-cluster

$x_i$  - Content of element  $i$  in matrix

$f_{ccl}$  - Co-cluster phase fraction

variable	value
precipitates strength	
TAU_CO_CLUSTS*	
TAU_CO_CLUST\$CL_MGSI_P0	1,99827e+07

# Precipitation strengthening, $\sigma_{prec}$

- 2 alternative models available
  - Size distribution dependent strengthening
  - Co-cluster strengthening

- $\tau_{prec} \rightarrow \sigma_{prec}$

$$\sigma_{prec} = M_T \tau_{prec}$$

$M_T$  - Taylor factor

# Precipitation strengthening, $\sigma_{prec}$

- 2 alternative models available
  - Size distribution dependent strengthening
  - Co-cluster strengthening

$$\tau_{prec} \rightarrow \sigma_{prec}$$

$$\sigma_{prec} = M_T \tau_{prec}$$

$M_T$  - Taylor factor

variables	value
kinetics: pd strength	
TSIGMA_PREC*	
TSIGMA_PREC\$nickelmatrix	1.23579e+09

category: kinetics: pd strength  
 expression: TSIGMA\_PREC\$nickelmatrix  
 legal unit qualifiers: \*none\*  
 -> total yield strength contribution from precipitates

# Mechanical threshold, $\sigma_0$


- Yield stress at temperature 0 K
- Thermal & athermal contributions

$$\sigma_{ath} = \sigma_i + \sigma_{gb} + \sigma_{sgb} + \sigma_{ss} + \sigma_{prec}$$

$$\sigma_{th} = \sigma_{disl}$$

$$\sigma_0 = f(\sigma_i, \sigma_{disl}, \sigma_{gb}, \sigma_{sgb}, \sigma_{ss}, \sigma_{prec})$$

$$\sigma_0 = \sigma_{ath} + \sigma_{th}$$



Thank you for your attention !



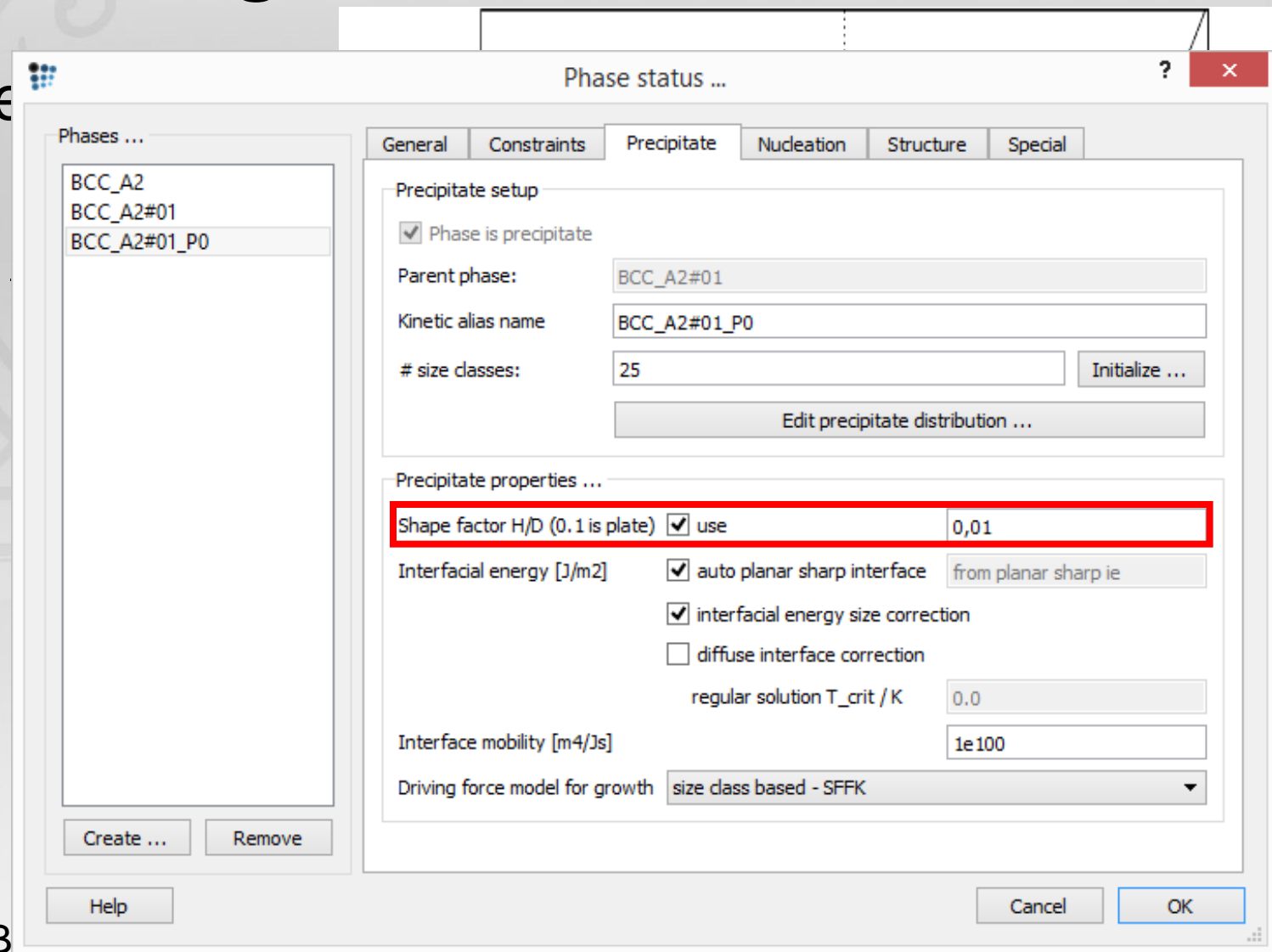
# Precipitation hardening

## Shape factor influence

$$L_S = K \left( \sqrt{\frac{\ln 3}{2\pi \sum_{class} N_{V,class} r_{m,class}^2}} + 4r_{ss}^2 \right)$$

$$K = h^{1/6} \left( \frac{2 + h^2}{3} \right)^{-1/4}$$

$h$  - Shape factor



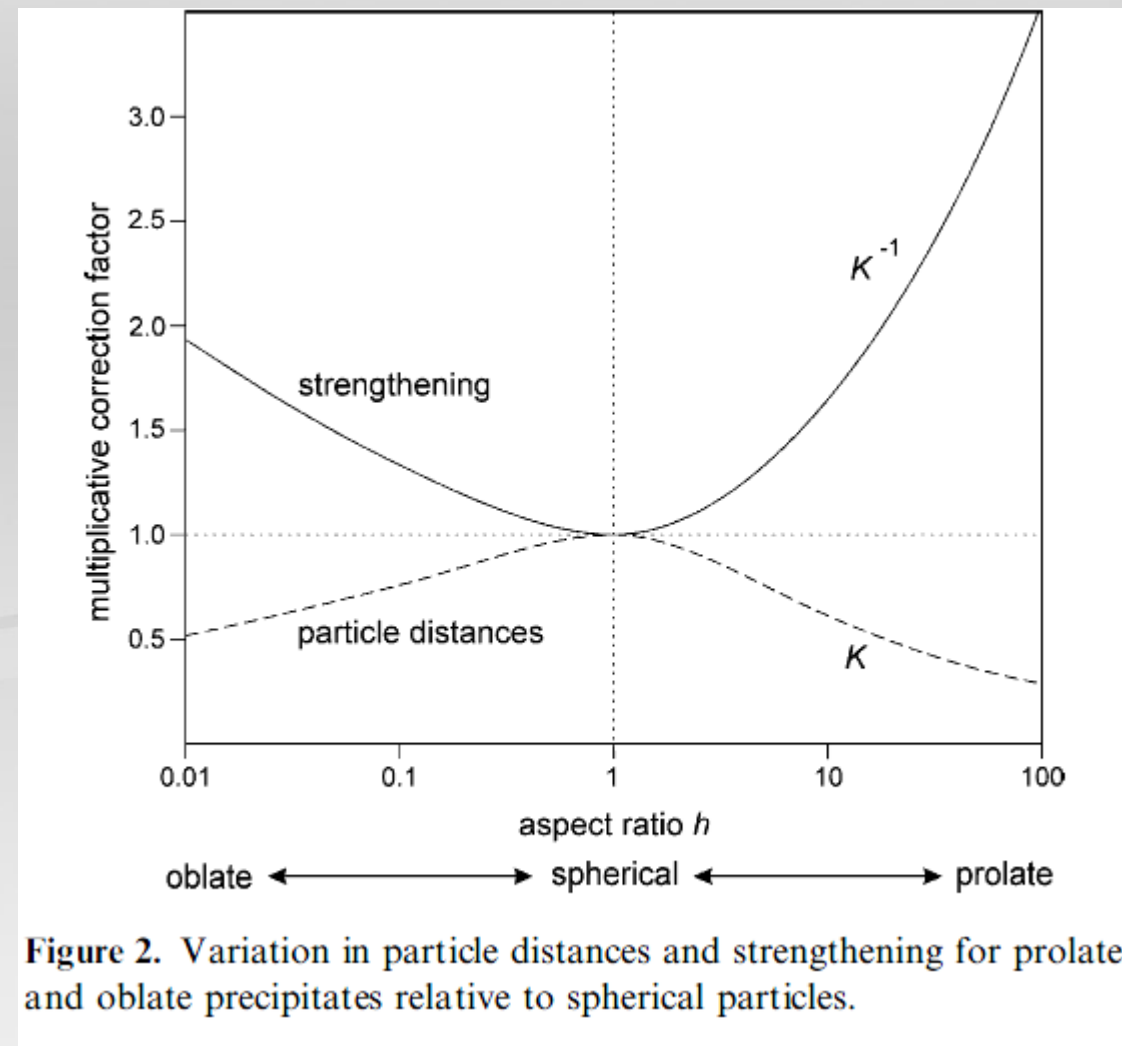
# Precipitation hardening

Shape factor influence on  $L_S$

$$L_S = K \left( \sqrt{\frac{\ln 3}{2\pi \sum_{class} N_{V,class} r_{m,class}^2} + 4r_{ss}^2} - 2r_{ss} \right)$$

$$K = h^{1/6} \left( \frac{2 + h^2}{3} \right)^{-1/4}$$

$h$  - Shape factor



# Precipitation hardening

Shape factor influence on  $L_S$

$$L_S = K \left( \sqrt{\frac{\ln 3}{2\pi \sum_{class} N_{V,class} r_{m,class}^2} + 4r_{ss}^2} - 2r_{ss} \right)$$

$$K = h^{1/6} \left( \frac{2 + h^2}{3} \right)^{-1/4}$$

$h$  - Shape factor

variables	value
kinetics: precipitates	
L_MEAN_2D\$*	
L_MEAN_2D\$GAMMA_PRIME_P0	2.02552e-08

category: kinetics: precipitates  
 expression: L\_MEAN\_2D\$GAMMA\_PRIME\_P0  
 legal unit qualifiers: \*none\*  
 -> mean distance between randomly distributed precipitates on a single plane (2-dimensional)

# Precipitation hardening

- Shearable particles – (e.g. coherency effect)

$$\tau_{coh,weak} = \frac{f(\theta)}{L_S} \left( \frac{G^3 \varepsilon^3 r_{eq}^3 b}{27T_{weak}} \right)^{1/2} h$$

$$r_{eq,edge,sh} = \left[ \frac{h^{2/3}}{3} \left( \sqrt{\frac{3}{2+h^2}} + 2\sqrt{\frac{6}{1+5h^2}} \right) \right] \frac{\pi}{4} r_m$$

$$r_{eq,screw,sh} = \left[ \frac{h^{2/3}}{3} \left( \frac{1}{h} + 2\sqrt{\frac{2}{1+h^2}} \right) \right] \frac{\pi}{4} r_m$$

$$r_{eq} = \left( P_{edge} r_{eq,edge,sh} + P_{screw} r_{eq,screw,sh} \right)$$

$r_{eq,edge,sh}$  - Equivalent radius for edge disl.

$r_{eq,screw,sh}$  - Equivalent radius for screw disl.

# Precipitation hardening

- Shearable particles – Coherency effect for non-spherical particles
  - Strong particles

$$\tau_{coh,strong} = \frac{(1.1101 \cos^2 \theta + 2.1488 \sin^2 \theta)}{L_s} \left( \frac{T_{strong}^3 G \epsilon r_m}{b^3} \right)^{1/4} K \quad K = h^{1/6} \left( \frac{2 + h^2}{3} \right)^{-1/4}$$

- Weak particles

$$\tau_{coh,weak} = \frac{(2.7310 \cos^2 \theta + 3.4736 \sin^2 \theta)}{L_s} \left( \frac{G^3 \epsilon^3 r_{eq}^3 b}{27 T_{weak}} \right)^{1/2} h \quad , \text{if } h \leq \frac{(1.3416 \cos^2 \theta + 4.1127 \sin^2 \theta)}{(2.7310 \cos^2 \theta + 3.4736 \sin^2 \theta)}$$

# Acknowledgments

- Mohammed R. Ahmadi
- Yao Shan



Thank you for  
your attention!

